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# **3D Hierarchical and porous layered double hydroxide structures: an overview of synthesis methods and applications**

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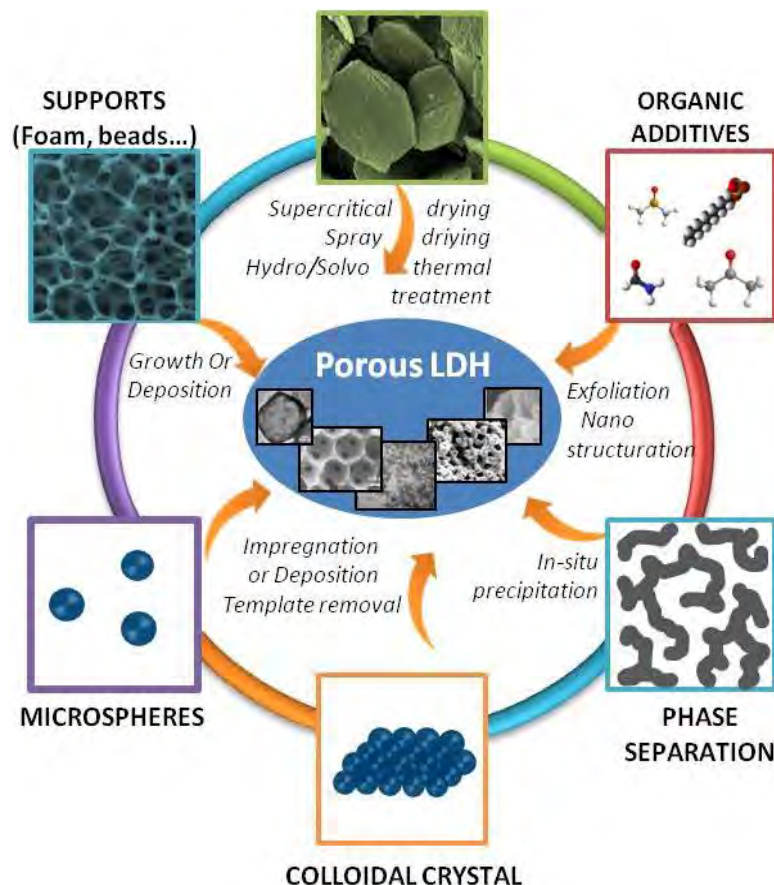
## **Abstract**

Nanostructured layered double hydroxide (LDH) materials with unique diffusion properties, large surface area along with desired functionalities have recently been produced for a number of well-established and advanced fields of applications. In this review, we describe and discuss the main synthetic methods that have been reported for the fabrication of porous LDH with tailored chemical composition and porosity. The efficiency of soft and hard templating approaches is particularly reviewed. A special emphasis is put on the microstructure and porosity of the materials according to the synthetic method involved. Finally, the performance enhancement of the materials due to the presence of porosity, especially macroporosity, in applications such as pollutant removal, catalysis and energy storage and conversion is overviewed.

Keywords: Nanostructured materials, layered double hydroxides, porosity, template synthesis, hierarchical structures

## 1. Introduction

Within the emergence of wide range of potential applications, layered double hydroxides (LDH) attracted during these last two decades increasing attention not only from the academic researchers but also from industrial community [1-4]. These materials are chosen for their unique bidimensional layers, their capacity to adsorb, to intercalate, and to immobilize species of interest. LDH defined by the general formula  $[M^{2+}_{1-x}M^{3+}_x(OH)_2]^{x+} [(A^{n-})_{x/n}, yH_2O]$  (usually abbreviated as  $M^{2+}M^{3+}$ -A, where  $M^{2+}$  and  $M^{3+}$  are respectively divalent and trivalent metals, and  $A^{n-}$  represents the interlayer anion compensating the positive charge of the metal hydroxide layers), are synthetic layered compounds with positively charged brucite-like layers of mixed metal hydroxides separated by interlayer hydrated anions. The main intrinsic properties of this class of materials are their adsorption behavior, related to their high layer charge density, their anionic exchange capacity in the range of 200-400 mEq/100g, and swelling abilities. Moreover, LDH present the advantage to be easily prepared in laboratory, through economically viable soft chemical processes, in terms of pressure, temperature and pH. By finely tuning the synthetic parameters, a huge variety of chemical composition involving diverse metal cations and interlayer anions can be achieved. It is noteworthy that the formation of LDH is in competition with those of related hydroxide or hydrated oxides. Thermodynamic studies evidenced that LDH precipitation is favored due to their greater stability compared to the corresponding simple hydroxides [5]. However, usually peculiar attention should be paid to avoid precipitation of side phases and contamination of carbonate anion. Although fine tuning of the porosity is required to improve the LDH performance in various applications involving reactions at the interfaces such as adsorption, catalysis, energy conversion, it still remains considerably challenging. While many reviews, journal special volumes, and book chapters detail the general LDH preparation and applications [2-4,6,7], the present one will focus mainly on 3D hierarchical and porous LDH materials using examples of the recent literatures from their elaboration to the features of porous LDH which enhance the LDH efficiency in various fields of applications. In this review, we will first outline how synthetic strategies based on soft and hard templating approaches can be used to prepare LDH porous materials as summarized in Scheme 1. The subsequent section will illustrate how such porosity control can be used advantageously in LDH potential applications, such as adsorption, catalysis, energy production, storage and conversion.



**Scheme 1** Schematic representation of the main synthetic approaches to prepare porous LDH

## 2. Synthesis method towards porous LDH

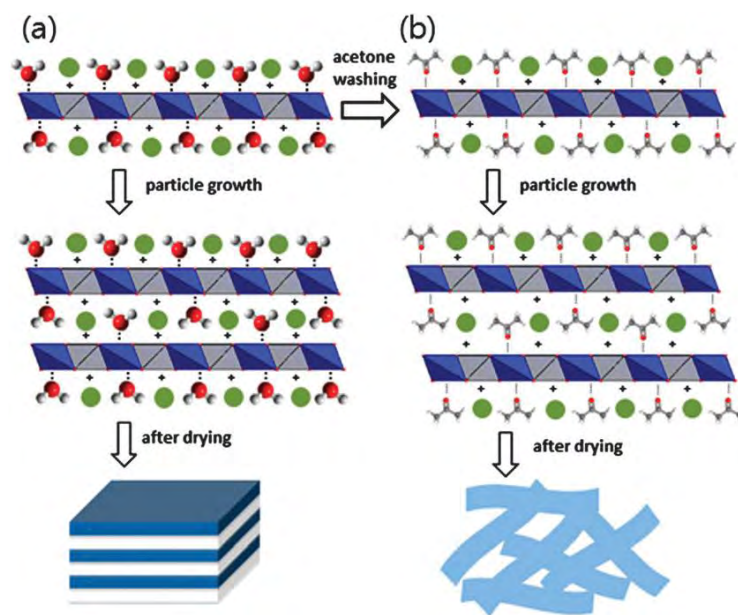
### 2.1 Textual porosity as a result of aggregated crystals.

The most primitive strategy towards porous LDH relies on the porosity among aggregated LDH crystals. The so-called textual porosity is introduced as a result of interstices of aggregated crystals. Even though the pore size distribution obtained thereby is broad and the pores are not well-organized, textual pores can increase surface area and enhance the accessibility to crystal surface.

LDH with the textual porosity are synthesized by an alkalization reaction from a starting mixture of metal salts dissolved in an aqueous solvent. Typical examples of synthesis methods are reviewed below. Mg-Al LDHs with textual porosity were demonstrated to form via coprecipitation reactions. NaOH/Na<sub>2</sub>CO<sub>3</sub> [8,9] or urea [10,11] works as an alkalization agent to precipitate LDH crystals from MgCl<sub>2</sub> · 6H<sub>2</sub>O and AlCl<sub>3</sub> · 6H<sub>2</sub>O at a relatively low temperature below 100 °C. Also, there has been reports on the synthesis of Ni-Al LDH [12], Co-Ni-Al LDH [13], Li-Al LDH [14] with textual porosity via hydrothermal and solvothermal methods, where high temperature and high pressure afford to yield LDH with a

relatively high purity. There have been some reports on the preparation of LDH hollow microspheres by optimizing crystallization, dissolution, and recrystallization of constituent crystals [15-17]. Activating the reaction steps of nucleation, crystal growth, and aggregation is a key to tune the textural porosity, and for this purpose, using sonication and/or adding ethylene glycol and glycine as a chelating agent have been employed [9,18,19]. The use of chelating agent also plays an important role to control structure and size of each crystals as well as the stability of crystals [20,21]. As a result, higher structural ordering is achieved in the presence of organic additives. Detailed discussion on this point is summarized in the section of 2.3 *Soft Templating*. LDH with a textural property is usually obtained as powders, whereas also it can be prepared on substrates, such as graphene and metal plates, through inhomogeneous nucleation [22,23]. Ni foil, Al sheet, porous anodic alumina aluminum substrate, MXene nanosheets were successfully used to support *in-situ* LDH growth [24-29] by simple immersion into a metal salt solution in presence of precipitating agent (NaOH, NH<sub>4</sub>OH, hexamethylenetetramine (HMT), urea...). Microstructures of LDH films can be easily tailored by the ion concentrations, the crystallization time and temperature. Zhang et al. further evidenced that during anion exchange reaction in presence of laurate anions, hollow hemispherical protrusions are generated at the film surface attributed to air bubble template mechanism [30].

Assembling exfoliated LDH nanosheets is another promising way to form textural porosity with a higher specific surface area compared to the pristine LDH. Although restacking of exfoliated nanosheets had been a critical drawback to achieve high porosity starting from exfoliated LDH nanosheets, O'Hare group recently developed a technique to avoid restacking with a technique named as the Aqueous Miscible Organic Solvent Treatment (AMOST) (Figure 1) [31,32]. This process forms a disordered card-house structure of LDH with a high porosity, achieving surface areas of 365 m<sup>2</sup>/g and 458.6 m<sup>2</sup>/g for Mg/Al-CO<sub>3</sub> [31] and Zn<sub>2</sub>Al-borate LDH [32], respectively. Such a technique with exfoliated nanosheets allows to functionalize LDH as well as to introduce porosity, leading to enhance adsorption [33], mechanical [34], optical [35], magneto-optical [36,37], and electrochemical [38] properties.



**Figure 1** Proposed mechanism for the formation of (a) conventional LDH and (b) highly dispersed LDH by the AMOST method. Reproduced from Ref. [32] with permission from The Royal Society of Chemistry

## 2.2 Post-treatment

Different post-synthetic treatments can be applied to pre-synthesized LDHs to control the particle aggregation, allowing to tune the LDH textural properties and morphologies. Spray drying has been used in order to prepare spherical microsphere with a diameter in the micron range formed by LDH nanoparticle aggregation. Typically, a LDH colloidal suspension obtained either by polyol method [39] or separated nucleation and aging steps [40] is sprayed to form an aerosol which leads upon solvent evaporation to spherical microparticles [41,42]. Spherical morphology is maintained during anion exchange and calcination, allowing the preparation of hybrid LDH and mixed metal oxide microspheres. These latter were efficiently involved in photocatalytic degradation [43] and used as sacrificial template for chemical vapor deposition (CVD) growth of graphene and porous graphene microsphere formation [44]. In limiting the capillary forces which occur during drying, supercritical drying is well-known as an efficient process to produce highly porous aerogels [45]. LDH aerogels were produced using LDH wet physical gels by applying water exchange with a non-aqueous solvent and subsequent CO<sub>2</sub> supercritical drying [46,47]. Even if monoliths are not systematically obtained, such treatment induces a net increase of the mesoporosity compared to standard ambient drying. Post-treatments have been also applied to LDH dispersed within a polymer matrix allowing to produce macroporous nanocomposites. A hierarchical porous LDH polyacrylamide nanocomposite with pores in both micro- and nanometer scales was prepared

after freeze-drying of the hydrogel [48]. Electrospinning was also applied to a poly- $\epsilon$ -caprolactone LDH nanocomposite to fabricate fibrous scaffolds [49]. LDH based nanocomposite foams were also reported involving *in-situ* polymerization process of polyurethane [50], and a high pressure CO<sub>2</sub> dissolution foaming process [51] of polystyrene, poly(styrene-co-acrylonitrile), and poly(methyl methacrylate). The *in-situ* coprecipitation of CoAl-LDH particles in the presence of exfoliated graphene oxide led to LDH nanosheet supported on graphene aerogel constructed by the physical cross-links between graphene sheets, via a one-step hydrothermal treatment assisted by a freeze drying process [52].

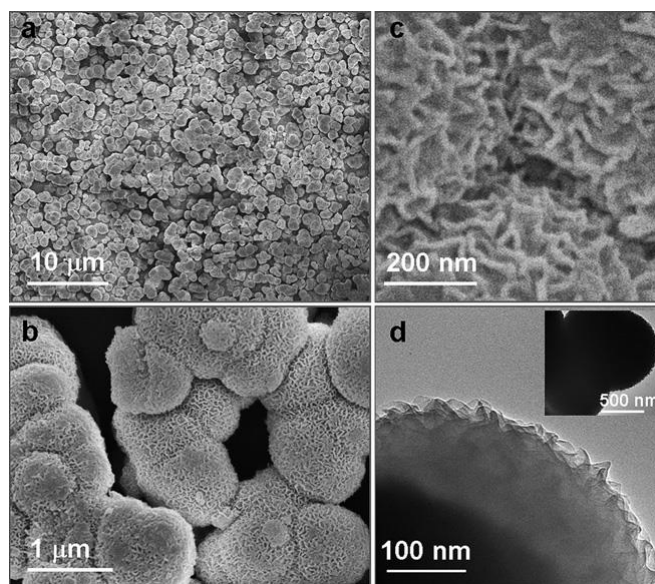
### 2.3 Soft templating

Soft-templating has been widely adopted to prepare mesoporous materials in alkoxide-derived sol-gel systems, where micelles of amphiphilic organic molecules (surfactants and water soluble polymers) are used to template well-defined mesoporous structures. Cooperative self-assembly between organic templates and inorganic precursors yield organized architecture and subsequent removal of organic templates by extraction or calcination create well-defined porosity. The structure of the mesophase depends on the packing properties of the surfactant molecules, and thereby structure of mesopores obtained by soft-templating is highly tunable by the nature of surfactant and composition of the starting mixture [53].

Considering that micelles represent colloidal dispersions with a particle size normally within 5 to 100 nm range [54], building blocks comprising mesowall are required to be small enough in nm range. The size of crystalline nanobuilding blocks used so far for the successful fabrication of well-ordered mesoporous structures is indeed less than 4-6 nm when pluronic surfactants (poloxamers) are used as soft templates [21,55,56]. For this reason, there are limited number of reports on the simultaneous achievement of well-defined periodic ordering of mesophase and crystallization of mesowall, such as organosilica [57], aluminosilicate [58], CeO<sub>2</sub> [56],  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> [59], and TiO<sub>2</sub> [60]. This requirement on the size of crystalline nanobuilding block imposes a primal challenge towards porous LDH via soft-templating because rapid crystallization kinetics of metal hydroxides, which typically form micron-scale crystals with a high crystallinity, prevents assembling them into an ordered mesophase. In addition, previous reports on mesoporous materials with crystalline wall typically involve a calcination step to crystallize precursory amorphous mesowall, whereas it cannot be applied to access to mesoporous hydroxides due to their lower thermal stability.

To date, there are various reports on the synthesis of porous LDH in the presence of

surfactant micelles. Gunawan et al reported the preparation of coral-like porous Mg-Al LDH microspheres in ethylene glycol/methanol/sodium dodecyl sulfate system (SDS) (Figure 2)[61].



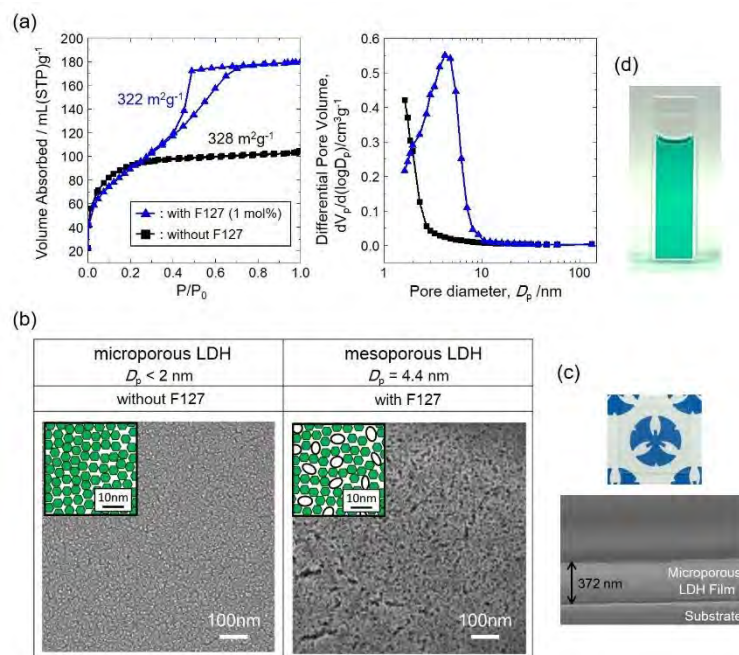
**Figure 2** SEM and TEM images of SDS-intercalated Mg-Al LDH prepared in a mixture of EG and methanol of 1: 1 volumetric ratio at 150 °C for 18 h. Reproduced from Ref. [61] with permission from The Royal Society of Chemistry

A flower-like Mg-Al LDH porous microstructure was demonstrated with SDS by Zhang et al and Sun et al.[62,63]. Shao et al reported the preparation of Mg-Fe LDH microspheres with the morphologies of hollow, yolk-shell, and solid interior structure by a hydrothermal reaction in NaOH aqueous system with SDS additive [64]. Using SDS as structure directing agent allows to change interlayer distance as well as macro-morphology (aggregated state) of LDH crystals [62,63]. Additionally, SDS anions can tune zeta potential of LDH nanosheets, affording to construct composites, such as graphene/LDH nanosheets [65].

Surfactant additives for LDH structuration that are in most cases SDS, apparently work in the different way from well-known “soft-temple” in alkoxide-derived systems. Pore size of LDHs are typically far larger compared to the size of micelles, and the formation of well-defined mesoporosity originated from surfactant micelles has not been clearly confirmed by characterization means of small angle X-ray scattering (SAXS) and N<sub>2</sub> adsorption. The organic surfactants and amphiphilic polymers rather work as nucleation sites [62,63] and/or an aggregation directing agent in macroscale [61]. Recently, successful soft-templating towards mesoporous LDH with using a nonionic surfactant (Pluronic F127) was achieved by limiting crystal growth and controlling aggregation [20]. Cooperative self-assembly between pluronic F127 and LDH nanocrystals with the size as small as 8 nm was successfully

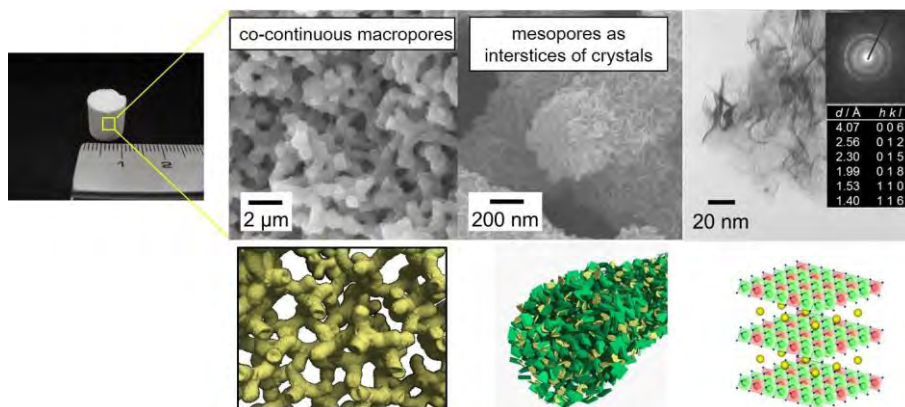


achieved (Figure 3). Further structural ordering of mesophase would be possible by reducing the size of crystals used as nanobuilding blocks.



**Figure 3** (a)  $N_2$  sorption isotherms and corresponding BJH pore size distributions, (b) FE-SEM images of soft-templated mesoporous LDH films with and without F127 additive. (c) Optical and SEM images of a microporous LDH thick film. (d) Suspension of LDH nanocrystals used for the coating. Adapted with permission from [20] Copyright (2016) American Chemical Society.

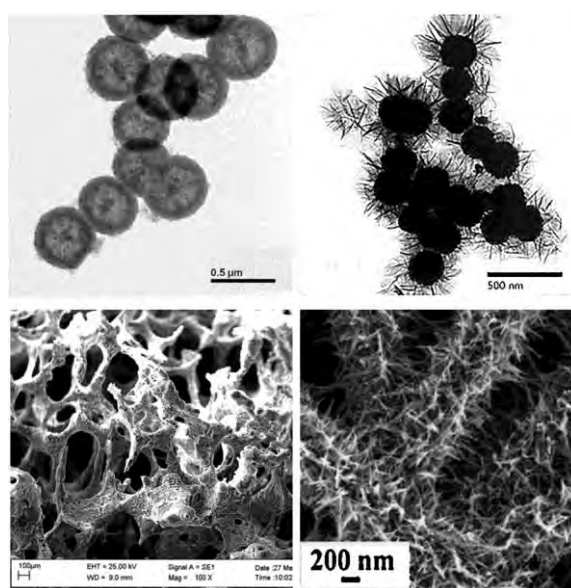
Another strategy towards porous LDH in the presence of a polymer is the alkalization reaction accompanied with phase separation. Starting from an aqueous mixture of metal salts, alkalization reaction induced by propylene oxide (PO) can crystallize LDH [66]. A homogeneous and fast pH increase is induced through the epoxide ring-opening reaction with the nucleophilic species [67]. Along with the pH increase, cationic species are consumed by the nucleation and growth of nanocrystals to form monolithic wet gel of hydroxides. The PO-mediated alkalization was demonstrated to be coupled with phase-separation phenomena, resulting in well-defined hierarchically porous structures [68]. Tokudome et al. demonstrated the synthesis of hierarchically porous Mg-Al LDH via PO-mediated alkalization accompanied with phase separation (Figure 4) [69]. Monolithic LDH materials with hierarchical pores in  $\mu\text{m}$  and nm ranges can be obtained via an aqueous one-pot reaction at as low as  $40^\circ\text{C}$ . This synthesis method exhibits compositional versatility and applicable to the preparation of  $M^{2+}_x\text{-Al}_x(\text{OH})_2\text{Cl}_x$  ( $M^{2+}$ :  $\text{Mg}^{2+}$ ,  $\text{Mn}^{2+}$ ,  $\text{Fe}^{2+}$ ,  $\text{Co}^{2+}$ ,  $\text{Ni}^{2+}$ ,  $\text{Cu}^{2+}$ ,  $\text{Zn}^{2+}$ ) with hierarchically porous structures [70].



**Figure 4** Hierarchically porous Mg-Al LDH monolith obtained through phase-separation accompanied with sol-gel transition [69].

## 2.4 Supported LDH synthesis

An interesting approach growing quite rapidly to produce LDH based hierarchical architectures is the use of 3D supports (Figure 5). Thanks to their positively charged surface covered by hydroxyl groups, LDH can easily interact with many kind of materials. As previously described in the case of 2D substrates, this approach allows a preferential orientation of LDH particles [25,29,71] and the preservation of morphology and/or macroporosity of the modified support.



**Figure 5** LDH supported on hollow mesoporous silica [72],  $\text{Fe}_3\text{O}_4$  spheres [73], FeCr alloy foam [74] and carbon nanofibers [75]. Adapted with permission from [72]. Copyright (2016) American Chemical Society. Reprinted from [74,75], Copyright (2016), with permission from Elsevier. Reprinted from [73], Copyright (2016), with permission from John Wiley and Sons.

Two main synthetic strategies can be distinguished to achieve the LDH deposition on supports either by the *in-situ* coprecipitation of LDH over the surface or by the self-assembly of

performed LDH nanoparticles at the surface. Table 1 summarized the main methods reported for preparing LDH on various kinds of supports. In one hand, to perform the *in-situ* coprecipitation of LDH, well-known synthetic methods reported for standard LDH synthesis such as, classical coprecipitation in presence of basic agent, coprecipitation using homogenous precipitation (involving retardant base *i.e.* urea, HMT...), electrosynthesis, or induced hydrolysis, are usually carried out in presence of the structured supports. Hydrothermal or solvothermal treatments can be associated to promote the crystal growth and tune the morphology of LDH coating. Typically, a selected support is introduced in an aqueous solution of mixed metal salts containing divalent and trivalent metal cations. Then, the adequate pH range for the LDH formation by coprecipitation is reached either by NaOH addition, thermal decomposition of a retardant base, or nitrate reduction. For instance, Abushrenta et al. reported the preparation of hierarchical CoAl LDH arrays by coprecipitation using urea in the presence of a  $\text{Co}(\text{OH})_2$  supported on macroporous Ni foam. A subsequent alkaline etching induced a partial Al removal producing mesoporous LDH layers. Due to an enhancement of the charge transport and short ion diffusion, such hierarchical architecture appeared as promising supercapacitors [76]. It is noteworthy that the support can act as a metal cation precursor through partial hydrolysis and contributes to the final LDH chemical composition. This is the case, for instance in the presence of  $\text{Al}_2\text{O}_3$  spheres [77,78] or biomorphic aluminum based fibers [79-81]. The *in-situ* growth of LDH nanoparticles vertically on the 3D support is an interesting strategy compared to platelet stacking, leading to macroporous structure.

In another hand, pre-synthesized LDH nanoparticles or exfoliated LDH nanosheets can also be deposited on the 3D support surface taking advantages of the LDH positively charge layers allowing electrostatic interaction with negatively charged supports. In this approach LDH coating is achieved either by simple deposition, by layer-by-layer process or by electrophoretic deposition (Table 1).

Using a self-assembly approach, hollow mesoporous silica spheres were successfully coated with exfoliated CoAl LDH and graphene oxide by alternate adsorption of nanosheets and subsequent chemical reduction to generate  $\text{SiO}_2/\text{LDH}/\text{graphene}$  hierarchical structure [72]. Obviously, a large variety of supports were reported including oxide microspheres ( $\text{Al}_2\text{O}_3$ ,  $\text{Fe}_2\text{O}_3$ ,  $\text{Fe}_3\text{O}_4$ ,  $\text{MnO}_2$ ...), nanowires (CuO, Al, Ag), metal foams (FeCr alloy, Ni), macro/meso alumina and silica, minerals such as zeolites, sepiolite and vermiculite, carbon nanotubes, 3D carbon oxide foam, carbon fibers, polymer foams or electrospun fibers, cellulose fibers and hierarchical carbon obtained by biotemplate calcination. Interestingly the final composite

materials benefits from the support morphology and intrinsic properties that is especially the case in using magnetic or conductive support such as iron oxides, metallic foams or carbon.

**Table 1** 3D hierarchical nanostructured LDH synthesized using various supports

Method	Metal salts/ precursor composition	Precipitating agent	Conditions	Support	Ref	
Co-precipitation	Ni(NO <sub>3</sub> ) <sub>2</sub> , Fe(NO <sub>3</sub> ) <sub>3</sub>	NaOH/ Na <sub>2</sub> CO <sub>3</sub>	65°C/6h	MnO <sub>2</sub> spheres	[82]	
	Mg(NO <sub>3</sub> ) <sub>2</sub> Cu(NO <sub>3</sub> ) <sub>2</sub> , Al(NO <sub>3</sub> ) <sub>3</sub>	NaOH/Na <sub>2</sub> CO <sub>3</sub>	65°C/36h	Fe <sub>3</sub> O <sub>4</sub>	[83]	
	Mg(NO <sub>3</sub> ) <sub>2</sub> , 6H <sub>2</sub> O Al(NO <sub>3</sub> ) <sub>3</sub>	NaOH/Na <sub>2</sub> CO <sub>3</sub>	60°C/24h	Fe <sub>3</sub> O <sub>4</sub>	[73]	
	Mg(NO <sub>3</sub> ) <sub>2</sub> or Ni(NO <sub>3</sub> ) <sub>2</sub> , Al(NO <sub>3</sub> ) <sub>3</sub>	NaOH/Na <sub>2</sub> CO <sub>3</sub>	RT/ 15 min	Fe <sub>3</sub> O <sub>4</sub>	[84]	
	MgSO <sub>4</sub> Al <sub>2</sub> (SO <sub>4</sub> ) <sub>3</sub>	NaOH	RT	Ti discs	[85]	
	Ni(NO <sub>3</sub> ) <sub>2</sub> ;CoCl <sub>2</sub>	-	180°C/ 24h	Ni foam	[86]	
	Mg(NO <sub>3</sub> ) <sub>2</sub> ,Al(NO <sub>3</sub> ) <sub>3</sub>	NaOH, Na <sub>2</sub> CO <sub>3</sub>	2h, RT	Zeolite	[87]	
	MgCl <sub>2</sub> ;AlCl <sub>3</sub>	NaOH	4h, RT	Sepiolite	[88]	
	FeSO <sub>4</sub> .Fe <sub>2</sub> (SO <sub>4</sub> ) <sub>3</sub>	NaOH, Na <sub>2</sub> CO <sub>3</sub>	-	Sepiolite	[89]	
	Ni(NO <sub>3</sub> ) <sub>2</sub> ; Co(NO <sub>3</sub> ) <sub>2</sub> ; Mn(NO <sub>3</sub> ) <sub>2</sub>	NaOH, Na <sub>2</sub> CO <sub>3</sub>	RT, 24h	MWCNT	[90]	
	ZnCl <sub>2</sub> , AlCl <sub>3</sub>	NaOH	RT, 24h	Cellulose	[91]	
	Zn(NO <sub>3</sub> ) <sub>2</sub> , Al(NO <sub>3</sub> ) <sub>3</sub>	NaOH Urea	100°C, 2h	Cellulose	[92]	
	Ni(NO <sub>3</sub> ) <sub>2</sub> ;Al(NO <sub>3</sub> ) <sub>3</sub>	Urea	HT 95°C/ 24h	Ag nanowire	[93]	
	Retardant Base	Ni(NO <sub>3</sub> ) <sub>2</sub> ;Al(NO <sub>3</sub> ) <sub>3</sub>	Urea	HT 120°C/12h	Ni foam	[94]
		Co(NO <sub>3</sub> ) <sub>2</sub> ;Al(NO <sub>3</sub> ) <sub>3</sub>	Urea	HT 120°C/6h	Ni foam	[95]
Co(NO <sub>3</sub> ) <sub>2</sub> ; Al(NO <sub>3</sub> ) <sub>3</sub>		Urea	HT 100°C/ 24h	Co(OH) <sub>2</sub> / Ni foam	[76]	
Fe(NO <sub>3</sub> ) <sub>3</sub>		Urea	HT 100°C/ 8h	Co <sub>2</sub> (OH) <sub>2</sub> CO <sub>3</sub> /Ni foam	[96]	
Mg(NO <sub>3</sub> ) <sub>2</sub> ; Al(NO <sub>3</sub> ) <sub>3</sub>		Urea	HT 120°C/ 24h	Vermiculite	[97]	
Ni(NO <sub>3</sub> ) <sub>2</sub> ;Al(NO <sub>3</sub> ) <sub>3</sub>		Urea	HT 100°C/24h	GO nanocups	[98]	
MnCl <sub>2</sub> ; NiCl <sub>2</sub>		HMT	HT 90°C/6h	Graphene sponge	[99]	
Co(NO <sub>3</sub> ) <sub>2</sub> FeCl <sub>2</sub>		HMT/I <sub>2</sub>	HT 100°C/3h	Carbon fiber cloth	[100]	
Co(NO <sub>3</sub> ) <sub>2</sub> ; Al(NO <sub>3</sub> ) <sub>3</sub>		Urea	HT 90°C/ 6h	Carbon fibers	[101]	
Ni(NO <sub>3</sub> ) <sub>2</sub> ; Co(NO <sub>3</sub> ) <sub>2</sub>		Urea or HMT	80°C/ 6h	Electrospun carbon nanofibers	[75]	
Zn(NO <sub>3</sub> ) <sub>2</sub> ; Al(NO <sub>3</sub> ) <sub>3</sub>		Urea	HT 120°C/10h	ZnCo <sub>2</sub> O <sub>4</sub>	[102]	
Mg(NO <sub>3</sub> ) <sub>2</sub> ; Al(NO <sub>3</sub> ) <sub>3</sub>		Urea	HT 120°C/24h	Electrospun PVDF	[103]	
Mg(NO <sub>3</sub> ) <sub>2</sub> ; Al(NO <sub>3</sub> ) <sub>3</sub>		Urea	HT 100°C/ 8h	Electrospun PAN/PMMA	[104]	
Electrosynthesis		Co(NO <sub>3</sub> ) <sub>2</sub> ; Fe(SO <sub>4</sub> )	Nitrate reduction	-1.0V, 0- 200s	CuO@Cu nanowire	[105]
		Ni(NO <sub>3</sub> ) <sub>2</sub> ; Fe(NO <sub>3</sub> ) <sub>3</sub>	Nitrate reduction	-0.1 mA/cm <sup>2</sup> 50-400s	WO <sub>3</sub> nanorod	[106]
	Ni(NO <sub>3</sub> ) <sub>2</sub> Al(NO <sub>3</sub> ) <sub>3</sub>	Nitrate reduction	-0.9 V/ -1.2V 600-1800s	FeCr alloy foam	[74]	
	Mg(NO <sub>3</sub> ) <sub>2</sub> ; Al(NO <sub>3</sub> ) <sub>3</sub> Rh(NO <sub>3</sub> ) <sub>3</sub>	Nitrate reduction	-1.2 V 2000 s	FeCr alloy foam	[107]	
	Ni(NO <sub>3</sub> ) <sub>2</sub> ; Fe(NO <sub>3</sub> ) <sub>3</sub>	Nitrate reduction	-1.0 V, 300s	Ni foam	[108]	
	Ni(NO <sub>3</sub> ) <sub>2</sub> /Co(NO <sub>3</sub> ) <sub>2</sub> /LiNO <sub>3</sub> Fe(SO <sub>4</sub> )	Nitrate reduction	-1.0 V, <300 s	Ni foam	[109]	
	Ni(NO <sub>3</sub> ) <sub>2</sub> ;CoCl <sub>2</sub>	Nitrate reduction	-1.0 V, < 200 s	PPy nanowire/Ni foam	[110]	
Induced hydrolysis	Ni(NO <sub>3</sub> ) <sub>2</sub>	NH <sub>4</sub> OH	100°C/ 48h	Fe <sub>3</sub> O <sub>4</sub> @SiO <sub>2</sub> @AlO <sub>3</sub> OH	[111]	
	Co(NO <sub>3</sub> ) <sub>2</sub> ; Cu(NO <sub>3</sub> ) <sub>2</sub> ,	NH <sub>4</sub> OH	HT 120°C/ 18h	Al <sub>2</sub> O <sub>3</sub> microspheres	[77]	
	Mg(NO <sub>3</sub> ) <sub>2</sub>	HMT	HT 120°C/12h	Al <sub>2</sub> O <sub>3</sub> /carbon fibers	[112]	
	MgSO <sub>4</sub>	HMT	HT 120°C/10h	Al <sub>2</sub> O <sub>3</sub> /C microspheres	[113]	
	Mg(NO <sub>3</sub> ) <sub>2</sub>	Urea	HT 90°C/ 24h	θ-Al <sub>2</sub> O <sub>3</sub> spheres	[78]	
	Mg(NO <sub>3</sub> ) <sub>2</sub> ; Ni(NO <sub>3</sub> ) <sub>2</sub>	NH <sub>4</sub> NO <sub>3</sub> , NH <sub>3</sub>	150°C/ 12h	Al Wire	[114]	
	Ni(NO <sub>3</sub> ) <sub>2</sub> , Co porous coordination polymer	-	Ethanol reflux 1h	Melamine polymer foam	[115]	
	CaCl <sub>2</sub>	Urea	150°C/3d	Mesoporous alumina	[116]	
	Mg(NO <sub>3</sub> ) <sub>2</sub> ;Ni(NO <sub>3</sub> ) <sub>2</sub> ; Co(NO <sub>3</sub> ) <sub>2</sub>	Urea	120°C/ 24h	Macroporous γ-Al <sub>2</sub> O <sub>3</sub>	[117]	
	Mg(NO <sub>3</sub> ) <sub>2</sub> ;Al(NO <sub>3</sub> ) <sub>3</sub>	-	600°C/6h HT rehydration	SBA-15	[118]	

	Mg(MeO) <sub>2</sub>	-	450°C/15h HT 125°C/21h	Meso-macro SBA-15	Al-	[119]
	MgAl-, ZnAl-, NiAl-	HMT/Urea	75°C-140°C 10h	Biotemplated Al <sub>2</sub> O <sub>3</sub>		[80,120, 121]
	Ni(NO <sub>3</sub> ) <sub>2</sub>	Urea	HT 100°C/24h	AlOOH/ Carbon NS		[79]
	Ni(NO <sub>3</sub> ) <sub>2</sub>	NH <sub>4</sub> OH	75°C/20h	AlOOH/C cloth		[122]
<b>Alternate deposition</b>	Co <sub>2</sub> Al- and Graphene oxide		Formamide	Hollow mesoporous SiO <sub>2</sub> spheres		[72]
<b>Electrophoresis</b>	MgAl-	-	1V- 20V	Tubular $\alpha$ -alumina		[123]
	Mg <sub>3</sub> Al- Ca <sub>3</sub> Al-	-	40V – 60 min	Activated carbon fiber cloth		[124]
<b>Layer by Layer</b>	MgAl- NiAl-	-	Alginate and chitosan	Polyurethane foam		[125]

## 2.5 Use of sacrificial template

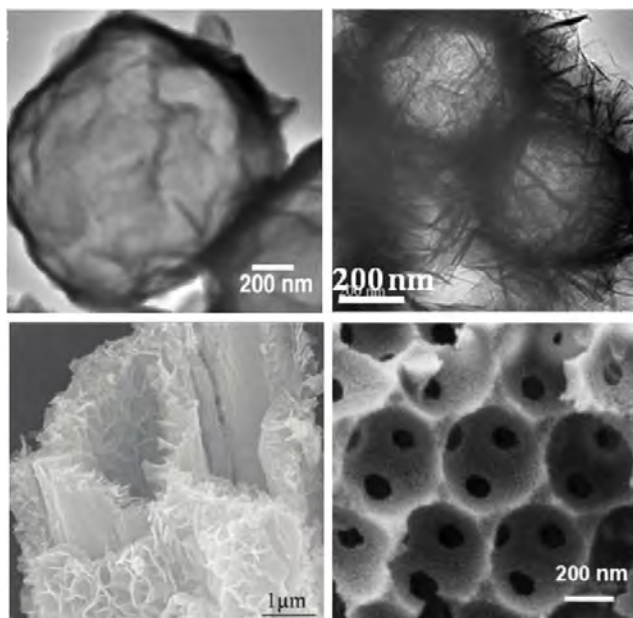
An alternative strategy to generate porous LDH is to use a sacrificial template which can be easily eliminated in a subsequent step. Usually in the first step, the LDH phase is associated to the sacrificial template following similar strategies as previously described for supported LDH. In the second step, the support is eliminated by dissolution or combustion leading to LDH porous structure. The different sacrificial templates and the LDH deposition methods used are listed in Table 2.

**Table 2** Porous LDH obtained by sacrificial templating approach.

Sacrificial template	LDH composition	Method	Removal technique	Ref
SiO <sub>2</sub>	MgAl-, NiAl	Induced hydrolysis	Basic dissolution	[126]
SiO <sub>2</sub>	MgAl-	<i>In-situ</i> coprecipitation	pH 11, 40°C	[127]
Hollow SiO <sub>2</sub>	CoAl-	Layer-by-layer	Extraction	[72]
Carbon nanospheres	MgAl-	Self-assembly	Calcination 500°C	[128]
Carbon nanosphere	NiAl-	<i>In-situ</i> template formation	Calcination	
Carbon Hollow microsphere/ SiO <sub>2</sub>	NiAl-	Induced hydrolysis	Etch process	[129]
$\gamma$ -AlO(OH)/Carbon fiber	MgAl-	Induced hydrolysis	Calcination 500°C 1h	[130]
PS spheres	CoFe-	Self-assembly	Calcination 700°C 4h	[131]
PS spheres	NiAl- MgAl-	Layer-by-layer	Calcination 480°C 4h	[132]
PS array	NiAl-	Nanoparticles infiltration	Dissolution	[20]
PS array	NiAl-	Electrosynthesis	Dissolution	[133, 134]
PS array	MgAl-	Successive impregnation	Calcination or dissolution	[135- 141]
Legumes	ZnAl-	Induced hydrolysis	Calcination 500°C/6h	[142]

An elegant alternative to produce porous networks was reported by Xiang et al., where the LDH crystallization and glucose carbonization is simultaneously induced under hydrothermal conditions [143]. Carbon nanospheres are formed conjointly with LDH and removed from the composite by thermal decomposition to generate mesoporous mixed metal oxides.

By this approach, the preparation of hollow LDH spheres [126,127,129,132,144], hollow LDH nanofibers [120,130,142] and 3D macro/mesoporous frameworks [134,135,137-139,141] have been reported (Figure 6). It should be remained that when sacrificial template is removed by calcination at a moderate temperature, layered mixed oxide is thermally crystallized in a certain extent with pristine LDH structure [2].



**Figure 6** Hollow spheres, hollow fibers and 3D ordered macroporous LDH structure obtained using sacrificial template. Reproduced from Ref. [132] with permission from the Royal Society of Chemistry. Adapted with permission from [126,139,142] Copyright (2016) American Chemical Society.

As illustrated in this part, synthetic efforts were focused on the preparation of porous LDH structure to enhance accessibility, diffusion within the materials and mass transport. Table 3 gathered the textural properties reported for some of the previously described morphologies demonstrating that both mesoporosity and macroporosity can be successfully created. Since playing on the pore networks was reported as an efficient approach to improve the material performances, the advantages/effects of 3D hierarchical and porous LDH nanostructure in various fields will be detailed in the following part.

**Table 3** Summary of textural properties of various porous LDHs.

Method	Composition	Morphology	Surface area (m <sup>2</sup> /g)	Macropores	Ref
<b>Urea hydrolysis</b>	ZnAlCO <sub>3</sub>	Hollow spheres	64	ND*	14
<b>Ethylene glycol</b>	MgAlCO <sub>3</sub>	Microspheres	165	ND	9
<b>Organic solvent treatment</b>	MgAl-CO <sub>3</sub>	Aggregated particles	365	ND	31,32
<b>Spray dried</b>	MgAlCO <sub>3</sub>	Microspheres	-	87.9 nm	42
	NiAlAcetate		72	ND	41
	MgAl-CO <sub>3</sub>		48	ND	44
	ZnAlCO <sub>3</sub>		103	ND	43
<b>CO<sub>2</sub> supercritical drying</b>	MgAlCO <sub>3</sub>	Aerogel	305	ND	47
	NiAlCO <sub>3</sub>				
<b>Pluronic F127</b>	NiAlCl	Mesoporous film	322	ND	20
<b>Phase separation</b>	MgAlCl	Hierarchically porous monolith	238 <sup>1</sup>	0.52 mm	69
<b>Spherical template</b>	MgAl-Ibuprofen	Hollow spheres	54	ND	128
<b>Inverse opals</b>	MgAlCO <sub>3</sub>	3DOM <sup>2</sup>	42	0.64 mm	139

\*ND : not determined <sup>1</sup>Sample calcined at 500°C <sup>2</sup>3-dimensionally ordered macroporosity

### 3. Field of applications of porous LDH

Porous LDH have been promising materials for various specific applications, taking advantage of high surface area and rapid molecular diffusion through interconnected pores (channels), as well as pristine surface properties of LDH crystals, such as hydrophilicity, positive electric-charge, and intercalation/deintercalation capability. Typical application fields of porous LDHs are briefly reviewed in this section.

### 3.1 Adsorption and removal for purification and environmental purposes

**Table 4** Adsorption on porous LDH

Adsorbate	LDH composition	Porosity of synthesized LDH	Adsorption capacity	Enhancement (vs standard LDH)	Ref
Orange II	MgAl-	Inverse opal	4.34 mmol·g <sup>-1</sup>	× 8.5	[137]
methyl orange	ZnAl-	Hierarchical textural porosity	248 mg·g <sup>-1</sup>	× 1.4	[15]
methyl orange	NiAl-	Hollow nanowires	210 mg·g <sup>-1</sup>	–	[16]
pyranine	MgAl-	Cocontinuous Hierarchically porous	1.8 mmol·g <sup>-1</sup>	× 3.5	[69]
Congo red	MgAl-	Hierarchical textural	447 mg·g <sup>-1</sup>	Comparable to activated carbon	[130]
Congo red	Ni/Mg/Al-	Flower-like hollow microspheres	1250 mg·g <sup>-1</sup>	× 2.1 of MgAl-LDH flakes	[146]
Cr(VI)			103 mg·g <sup>-1</sup>	× 6.7 of ZIF-67 microcrystals	
Phosphate	Zn/Al-	Hollow microspheres	232 mg·g <sup>-1</sup>	–	[17]
F <sup>-</sup>	Li/Al-	Hierarchical textural porosity	159 mg·g <sup>-1</sup>	–	[14]
F <sup>-</sup>	Zn/Al-	LDH supported on cellulose	25.2 mg·g <sup>-1</sup>	× 2–4	[91]
S <sub>2</sub> O <sub>3</sub> <sup>-</sup>	Ni/Al-	Grown on Ni foam	209 mg·g <sup>-1</sup>	–	[94]

Table 4 shows the relationship between adsorption capacities and pore characteristics of porous LDH. Previous reports in the list successfully demonstrated advantages/improvements by the introduction of porosity into LDH materials. High loading of adsorbates on porous LDH has been demonstrated for anionic dye molecules, such as orange II [137], methyl orange [15,16], pyranine [69,145] and Congo-red [130], where various porous morphologies, including hollow spheres [15], hollow nanowires [16], 3D ordered macroporous [137], hierarchically-porous monolith [69], microtubes [130], and urchin-like particles [146], have been employed. The transport of molecules in porous LDH primarily depends on the macropores (macrochannels) and the introduction of larger macropores accelerates adsorption kinetics more progressively [145]. Whereas the introduction of mesopores, increases loading capacity at an equilibrium state [145]. There has been various reports on the application of porous LDH towards environmental purposes, targeting at harmful anionic species in waste water, such as phosphates [17], various oxyanions (SO<sub>4</sub><sup>2-</sup>, CrO<sub>4</sub><sup>2-</sup>, MoO<sub>4</sub><sup>2-</sup>, HVO<sub>4</sub><sup>2-</sup>) [70], and fluoride ion [14,91], thiosulfate [94], arsenic [89], and chromium(VI) oxyanion [97]. Porous



LDH have been also used for the extraction of phenolic compounds [135], and oil separation from oil-polluted water [104].

### 3.2 Energy production, storage and conversion

Highlighted electrochemical properties of porous LDH-base electrodes are listed in Table 5. LDH have been widely applied as electrode materials due to their high theoretical specific capacitance, high redox activity, low manufacturing cost, and environmental-friendly nature. The enhancement of capacitance of LDH electrodes has been achieved by the introduction of textured pores [12,23,100], and hierarchical porosity [76]. The capacitance of LDH-based electrodes was reported to be further enhanced by compositing LDH with carbon-based field of application that Ni foam can be used as Ni metal source as well as conductive substrate to produce a metal hydroxide electrode [148]. This technique allows one-step method to fabricate Ni-Al LDH electrode and is widely applied for the electrochemical applications of

**Table 5** Electrochemical properties of porous LDH-based electrodes

<b>Porous LDH employed</b>	<b>Highlighted properties</b>	<b>Ref</b>
Flowerlike hierarchically porous Ni/Al-LDH	* $C_s = 477 \text{ F} \cdot \text{g}^{-1}$ at $J = 12 \text{ A} \cdot \text{g}^{-1}$	[12]
3D porous NiAl-LDH grown on graphene	$C_s = 1256 \text{ F} \cdot \text{g}^{-1}$ at $J = 1 \text{ A} \cdot \text{g}^{-1}$ $C_s = 756 \text{ F} \cdot \text{g}^{-1}$ at $J = 6 \text{ A} \cdot \text{g}^{-1}$	[23]
CoFe-LDH nanosheets grown on carbon fiber	$C_s = 774 \text{ F} \cdot \text{g}^{-1}$ at $J = 1 \text{ A} \cdot \text{g}^{-1}$ , 91% retention of the initial $C_s$ after 5000 cycles at $J = 2 \text{ A} \cdot \text{g}^{-1}$	[100]
Co(OH) <sub>2</sub> @porous LDH grown on Ni foam	$C_s = 1734 \text{ F} \cdot \text{g}^{-1}$ at $J = 5 \text{ mA} \cdot \text{cm}^{-2}$ , 85% retention of the initial $C_s$ after 5000 cycles at $J = 5 \text{ mA} \cdot \text{cm}^{-2}$	[76]
CoAl-LDH/graphene oxide/ MnO <sub>2</sub> nanocomposite with mesoporosity	$C_s = 340\text{-}560 \text{ F} \cdot \text{g}^{-1}$ at $J = 2 \text{ A} \cdot \text{g}^{-1}$	[11]
NiCo <sub>2</sub> S <sub>4</sub> nanotube@NiMn-LDH/graphene sponge	$C_s = 1740$ and $1268 \text{ mF} \cdot \text{cm}^{-2}$ at $J = 1$ and $10 \text{ mA} \cdot \text{cm}^{-2}$ , respectively, 84.5% retention after 5000 cycles	[99]
3D porous CoAl-LDH/graphene aerogel	$C_s = 640$ and $305 \text{ F} \cdot \text{g}^{-1}$ at $J = 1$ and $20 \text{ A} \cdot \text{g}^{-1}$ , respectively, 97% retention after 10000 cycles	[52]
NiCo-LDH supported on Ni form	$C_s = 2682 \text{ F} \cdot \text{g}^{-1}$ at $J = 3 \text{ A} \cdot \text{g}^{-1}$ , energy density of $77.3 \text{ Wh} \cdot \text{kg}^{-1}$ at $623 \text{ W} \cdot \text{kg}^{-1}$	[86]

\* $C_s$ : specific capacitance;  $J$ : current density

LDH materials [86].

From different application aspects, noteworthy remarks and highlights on using porous LDH

in catalysis, bio, and others are summarized in Table 6 We will briefly go through in the following sections.

### ***3.3 Catalysis***

The introduction of porosity increases surface area and thereby enhances the catalytic activity of LDH-based materials. Some examples of the improvement of activity/selectivity in catalytic applications are listed in Table 6. For example, Halma et al reported that oxidation of heptane exhibits selectivity for the alcohol product over iron porphyrins accommodated in LDH with macroporous structure [140]. The authors explained that the reaction selectivity was impacted by the structure of LDH support with channel and micro-environments which can create a suitable structure for the access of substrates toward the reaction sites. Various catalytic reactions, such as alcohol oxidation [140], aldol condensation [118], ethanol electrooxidation [64], pollutant photodegradation [136] and photoelectrochemical water splitting [106], have been reported for porous LDH catalysts. Nanostructured LDH (nanoarray and nanoflakes) prepared on a Ni foam were also employed as electrocatalysts of oxygen evolution reaction (OER) [96,127,149]. A porous hydroxide showing OER activity was also prepared by selective etching of M(III) in hydroxide sheets [150]. In addition, mesoporous oxides with well-defined porosity have been used as a support to load catalytic LDH crystals; LDH precipitated on mesoporous AlOOH was used for selective hydrogenation [78] and as CO<sub>2</sub> adsorber after calcinations [116].

**Table 6** Summary of catalytic, bio-, and other applications of LDH

Application	Composition and role of LDH	Porosity of LDH	Highlights, remarks, comparison with standard LDH	Ref
Catalysis	Host material (MgAl-)	3D ordered macropores	Reaction selectivity imparted by the macroporosity	[140]
Catalysis	Electroactive material (MgFe-)	Hollow microspheres	Electrocatalytic oxidation of ethanol achieved in alkaline fuel cell	[64]
Photolysis	Host material (MgAl-)	3D ordered macropores	Enhanced reaction kinetics due to the macroporosity	[136]
Photoelectrochemical water splitting	Co-catalyst (NiFe-)	Nano LDH flakes on WO <sub>3</sub> nanorod arrays	Benefit for light absorption and migration of carriers,	[106]
Electrocatalytic water splitting	Precursor (NiGa-)	Porous nanosheets	Selective etching of LDH to prepare porous hydroxide and chalcogenides	[150]
Hydrogenation	Catalyst support (MgAl-)	LDH-modified porous alumina	Higher catalytic activity/selectivity compared to standard	[78]
Amperometric detection of H <sub>2</sub> O <sub>2</sub>	Host material (MgAl-)	Textural pores	Detection limit of 1.5×10 <sup>-8</sup> M, sensitivity of 37 A·M <sup>-1</sup> cm <sup>-2</sup>	[151]
Trypsin adsorption	Adsorbent (MgAl-)	Aerogel with textural porosity	20 times higher adsorption capacity than conventional LDH	[47]
Bovine serum albumin adsorption	Adsorbent (MgAl-)	Aerogel with hierarchical porosity	>14 times higher adsorption capacity than standard LDH	[46]
Controlled drug release	drug release media (MgAl-)	LDH on porous titania	Pores prevent premature detachment of the LDH coating.	[85]
Controlled drug release	Host material (Mg/Al-)	Hollow microspheres	Better dispersion in the liquid phase and higher surface area	[128]
Non-enzymatic glucose sensor	Redox mediator (NiAl-)	3D macroporous LDH on carbon cloth	14.13 mA mM <sup>-1</sup> ·cm <sup>-2</sup> , response time less than 1 s	[122]
Glucose biosensor	Enzyme support (NiAl-)	3D ordered macropores	×1.4 enhancement of sensitivity due to the macroporosity	[134]
Tissue engineering scaffold	Nanofiller (MgAl-)	Grown on 3D porous polymer	Improved the tensile strength, elongation, and proliferation and differentiation of cells	[49]
Li-ion battery	Electrode material ((Zn/Cu)Al-)	Interconnected nano-sheets with porosity	LDH-derived oxides exhibit higher capacity & better stability than pure ZnO	[22]
Photocatalysis	Adsorbent/catalyst (ZnAl-)	Textural porosity	Hierarchical structure improve the adsorption and photocatalytic properties	[102]
Fire suppression	Fire extinguishing agent (MgAl-)	Porous microspheres	Higher efficiency in suppressing gasoline pool fire	[9]
Superhydrophobicity	Microstructure (NiAl-)	Hexagonal micro-structures	Allow to immobilize lauric acid on rough surface to show contact angle of 163°	[24]
Electrochromic	Inverse opal (NiAl-)	Chromophore	Improvements in the electrochromic properties (×4 larger color change)	[133]
Antireflection coating	Porous film (MgAl-)	Switchable porous material	AR properties switchable by the reconstruction effect of LDH	[71]
NO <sub>x</sub> Sensing	Sensor (MgAl-)	Hierarchical flower-like LDH	Response in 1.3 s and recovery time of 30 s to 100 ppm NO <sub>x</sub>	[63]

### 3.4 Bio-applications

LDH have been investigated as biocompatible host materials due to the possibility of providing a suitable microenvironment for biomolecules, such as proteins and enzymes [151].

The introduction of porosity, especially large mesopores and/or macropores, can increase the

surface area which is accessible by large biomolecules with tens nm in size. High-density protein loading on porous LDH was reported for trypsin [47] and bovine serum albumin (BSA) [46,121], where enhancement of adsorption were reported as 20 times and 14 times compared to those for standard LDH, respectively (Table 6). There has been also various reports on loading and releasing an antibiotic [85], drug [128], enzyme [47], glucose biosensor development [122,134] and tissue-engineering scaffolds [49].

### **3.5 Others**

Porous LDH materials have been used as precursor for porous metal oxides. For example, porous ZnO/ZnAl<sub>2</sub>O<sub>4</sub> crystalized from Zn-Al LDH exhibit excellent cycling stability as an anode material for Li-ion batteries [22] and enhanced adsorption and photocatalytic properties toward Congo red [102]. Also, there are reports on the applications of porous LDH for fire suppression [9], designing superhydrophobic surface [24], electrochromic coating [133], erasable antireflective [71], and gas sensors [63]. For all these applications, the introduced porosity play important roles to generate respective properties (Table 6).

## **4. Concluding remarks**

Over the last decade, 3D hierarchical and porous LDH has gained an increasing interest allowing the preparation of nanostructured LDH materials which displayed real advantages for various applications such as adsorption for environmental purposes, energy production, storage and conversion, photo, electro- catalysis and biosciences. Even if their synthesis is often challenging, the large panel of LDH synthetic pathways using soft chemistry conditions allowed to successfully design various LDH systems. For a given nanostructure, porous and hierarchical LDH materials may provide multifunctional properties by acting as a passive or an active host material with high surface area allowing to accommodate other components within the nanostructure or at the wall surface. By this way, nanostructured multicomponent systems can be achieved opening new opportunity for the applications of LDH-based

materials thanks to the functionality of accommodated component as well as advantages of pristine LDH, such as a rapid mass or charge transport. Based on the state of the art, the authors strongly believe that in a near future, even more complex LDH based porous assemblies may be designed displaying well-defined architectures of highly-controllable size with a better level of understanding of the synthesis-structure-properties relationship. A method of preparing porous LDH monoliths displaying good mechanical and thermal properties will be also necessary to open the way to actual applications.

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