



# Investigations on the Development of Highly Active Titanium Oxide Photocatalysts and their Reactivity for the Oxidation of Organic Compounds

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**Investigations on the Development of  
Highly Active Titanium Oxide Photocatalysts  
and their Reactivity for the Oxidation of Organic Compounds**

(高活性な酸化チタン光触媒の開発と  
有機化合物の酸化反応における反応性に関する研究)

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Doctoral Thesis at Osaka Prefecture University

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## **Chapter 1**

### **General Introduction**

# **1. General Introduction**

## **1. Introduction**

Environmental pollution on a global scale as well as the lack of natural energy resources have drawn much attention to the vital need for ecologically clean chemical technologies as one of the most urgent challenges facing chemists today. Since the photosensitization effect of a TiO<sub>2</sub> electrode on water electrolysis was first discovered by Honda and Fujishima in 1972 [1], pollution-free photocatalysis by TiO<sub>2</sub> semiconductors has been widely studied in order to achieve the efficient conversion of clean solar energy into useful chemical energy such as hydrogen [2-9]. New systems and processes powered by clean solar energy will not only resolve energy issues caused by the exhaustion of fossil fuels but can also be applied for the abatement of environmental toxins. Along these lines, photocatalysts which can operate under visible and/or solar light irradiation have been strongly desired for applications in the purification and sustenance of our living environment.

Various studies have also been carried out on TiO<sub>2</sub> nano-particles as well as on various Ti-oxide based binary oxides such as TiO<sub>2</sub>/SiO<sub>2</sub>, TiO<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub>/B<sub>2</sub>O<sub>3</sub> [10-13]. In particular, we have found that the TiO<sub>2</sub> nano-particles of less than 10 nm show a

significant enhancement in photocatalytic reactivity under UV light irradiation. This phenomenon is due to an electronic modification of the TiO<sub>2</sub> nano-sized semiconductors as well as the close existence of the photo-formed electron and hole pairs and their efficient contribution to the photoreactions. These findings have provided new insights into the development of the highly dispersed transition-metal oxide species as single-site catalysts. Moreover, the application of an anchoring method enabled to prepare the molecular or cluster-sized photocatalysts on various supports such as SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, various zeolites and mesoporous materials. Highly dispersed Ti-, Cr-, Mo-oxide species incorporated within the cavities or frameworks of zeolites are especially interesting due to their unique local structures such as the four-fold coordinated species and efficient photocatalytic properties for the reduction of CO<sub>2</sub> with H<sub>2</sub>O, NO decomposition as well as the selective photoepoxidation of alkene with O<sub>2</sub>, as compared with semiconducting photocatalysts [14-24].

Recently, various air-cleaning systems equipped with TiO<sub>2</sub> photocatalysts and UV light sources that reduce volatile organic compounds (VOCs) which cause the so-called “sick house syndrome” are commercially available. However, the removal efficiency of air-cleaning systems still needs to be improved to be as simple and low-cost as possible for widespread applications. Although the deposition of small amounts of Pt particles on the

TiO<sub>2</sub> surface is known to enhance their photocatalytic reactivity [25-29], Pt is too costly for common use in home electrical appliances. Meanwhile, the hybridization of adsorbents such as zeolites or mesoporous materials [14-19, 30-34] with TiO<sub>2</sub> particles has been reported to show elevated photocatalytic reactivity. In the previous report [35], the TiO<sub>2</sub> nano-particles hybridized with siliceous zeolites prepared by an impregnation method as well as a simple mechanical blending method showed higher photocatalytic reactivity for the complete oxidation of gaseous acetaldehyde than TiO<sub>2</sub> catalysts since the siliceous zeolites can efficiently condense acetaldehyde thinly diffused in the gas phase and smoothly supply them onto the TiO<sub>2</sub> photocatalyst surfaces.

The conversion of solar light energy into renewable clean energy is also one of the most challenging research topics in science and technology. The sunlight including near-infrared, visible and ultraviolet light provide tremendous energy of ca. 87 - 308 kJ mol<sup>-1</sup> so that solar energy should be utilized as efficiently as possible [36-37, 60-64]. It will, thus, be of great importance to develop the effective systems able to convert abundant solar light energy into applicable and sustainable energy resources. At least two systems have been considered for the conversion of sunlight into other renewable energy sources: one is the design of solar cells to convert sunlight into electricity and the other is artificial photosynthesis for the conversion and storage of solar energy into safe



and useful chemical energy such as hydrogen. Although hydrogen is also the focus of much attention as a renewable clean energy alternative, at the moment, we do not have any highly efficient systems to produce hydrogen in an environmentally harmonious way without producing CO<sub>2</sub>. From this viewpoint, the photocatalytic or photoelectrochemical decomposition of water to produce hydrogen under solar light irradiation is now of utmost importance.

In this thesis, the development of highly functional Ti-oxide based photocatalysts, i.e., (i) the tetrahedral Ti-oxide species incorporated within the framework of zeolites and mesoporous materials as single-site photocatalysts; (ii) the TiO<sub>2</sub> nano-particles hybridized with hydrophobic zeolite adsorbents in practical applications for photocatalytic air-cleaning systems; and (iii) the TiO<sub>2</sub> thin films to photocatalytically decompose H<sub>2</sub>O into H<sub>2</sub> and O<sub>2</sub> under solar light irradiation will be discussed.

## **Chapter2**

The synthesis, characterization and photocatalytic performance of the TiO<sub>2</sub> nanopowders prepared by a flame-synthesis method were investigated. The photo-excited states of TiO<sub>2</sub> nanopowders under UV-light irradiation were directly observed by an in-situ NEXAFS (Near Edge X-ray Absorption Fine Structure) study. It was found that the anatase/rutile phase boundary works as an electron trapping site. By the combination of

TEM images, the enhancement of the photoreactivity of the TiO<sub>2</sub> nanopowders was attributed to both the changes in the particle shape and the existence of an anatase/rutile phase boundary, on which the excited electrons have long lifetimes and suppress the recombination of the photo-formed electrons and holes.

### **Chapter3**

TiO<sub>2</sub> photocatalysts were prepared by a multi-gelation method and the effect of the changes in the pH of the pH swing method, on the morphology of the TiO<sub>2</sub> particles was investigated. The photocatalytic properties of the TiO<sub>2</sub> nano-powder prepared by the controlled pH swing method were compared with the TiO<sub>2</sub> particles prepared without adjusting the pH value during the swing times. The photocatalytic performances of these TiO<sub>2</sub> nano-powders were investigated by comparing the photocatalytic degradation reaction of 2-propanol under UV light irradiation. The experimental results showed that the TiO<sub>2</sub> photocatalysts prepared without adjusting the pH showed better performance in controlling the important parameters of the catalysts such as the particle size, surface area, anatase/rutile phase ratio, the pore size as well as pore volume than the TiO<sub>2</sub> photocatalysts prepared by a controlled pH swing method.

### **Chapter4**

The photocatalytic oxidation of gaseous acetaldehyde with O<sub>2</sub> on the commercial TiO<sub>2</sub> nano-particles could be successfully enhanced by a simple mechanical blending with a

siliceous mordenite (MOR) zeolite, which showed highly hydrophobic surface. When the TiO<sub>2</sub> nano-particles of ca. 5 - 20 wt% were mixed with the MOR zeolite powders in an agate mortar for only 5 min, the blended TiO<sub>2</sub>/MOR samples showed higher photocatalytic reactivity as compared with the pure untreated TiO<sub>2</sub> nano-particles. Since the siliceous zeolite powders are highly transparent in UV-VIS light regions, the incident UV light is effectively irradiated onto the whole part of the TiO<sub>2</sub> nano-particles without any loss of light intensity. Furthermore, the hydrophobic MOR zeolite powders effectively adsorb the gaseous acetaldehyde molecules and supply them onto the surfaces of the blended TiO<sub>2</sub> nano-particles, resulting in the enhancement of the photocatalytic reactivity.

## **Chapter5**

TiO<sub>2</sub> thin film photocatalysts which could induce photoreactions under visible light irradiation were successfully developed in a single process by applying an ion engineering technique, i.e., a RF magnetron sputtering deposition method. The TiO<sub>2</sub> thin films prepared at higher than 773 K showed the efficient absorption of visible light, on the other hand, the TiO<sub>2</sub> thin films prepared at around 473 K was highly transparent. This clearly means that the optical properties of the TiO<sub>2</sub> thin films, which absorb not only UV but also visible light, can be controlled by the preparation temperatures of the RF magnetron

sputtering deposition method. These visible light responsive TiO<sub>2</sub> thin films were found to exhibit effective photocatalytic reactivity under visible light irradiation ( $\lambda > 450$  nm) at 275 K for the reductive decomposition of NO into N<sub>2</sub> and N<sub>2</sub>O. From various characterizations, the orderly aligned columnar TiO<sub>2</sub> crystals could be observed only for the visible light responsive TiO<sub>2</sub> thin films. This unique structural factor is expected to modify the electronic properties of TiO<sub>2</sub> semiconductor, enabling the efficient absorption of visible light.

## **Chapter6**

These conclusions obtained from the investigations covered in this thesis have been summarized in this chapter.

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## **Chapter 2**

**Suppressed Recombination of Electrons**

**And Holes And its role in the improvement of**

**the Photoreactivity of Flame-synthesized TiO<sub>2</sub> Nanopowders**

## **2. Suppressed Recombination of Electrons and Holes and its role in the improvement of the Photoreactivity of Flame-synthesized TiO<sub>2</sub> Nanopowders**

### **2.1. Introduction**

Titanium dioxide (TiO<sub>2</sub>) is well-known as an effective photo-functional material which exhibits photocatalytic properties for reactions in environmental purification, hydrogen production from water, and superhydrophilic properties under UV-light irradiation. Many researchers have, thus far, endeavoured to improve its photocatalytic efficiency by different synthetic techniques. TiO<sub>2</sub> photocatalysts have been fabricated using such methods as sol-gel, hydrothermal treatment, and other physical methods [1-3]. Among them, the flame synthesis approach is one of the most effective in preparing TiO<sub>2</sub> nanopowders with high photocatalytic performance [4-6]. In general, the flame synthesis of TiO<sub>2</sub> nanopowders is carried out at high temperatures above 900°C, affording TiO<sub>2</sub> nanoparticles with very high degrees of crystallinity. Therefore, no additional calcination at high temperature is necessary to improve the crystallinity [7-12]. This fabrication method results in a low degree of agglomeration of the TiO<sub>2</sub> nanoparticles, which is crucial to obtaining high dispersibility in liquid media as well as a large specific surface

area. Moreover, the properties of a synthesized TiO<sub>2</sub> nanopowder can be modified by post-treatments such as surface etching in acid solution [13] and heat-treatment at elevated temperatures in a controlled atmosphere [4, 14]. However, such post-treatments of a TiO<sub>2</sub> nanopowder are not always effective since the physico-chemical properties of TiO<sub>2</sub> nanopowders such as the lattice defects inside the bulk and anatase-to-rutile phase ratios are often controlled by the synthesis method.

Along these lines, we report on the effect of heat-treatment of TiO<sub>2</sub> nanopowders prepared by the flame-synthesis method. In particular, attention is focused on the relationship between the photo-excited structure and the photocatalytic performance of the TiO<sub>2</sub> nanopowders.

## **2.2. Experimental**

### **2.2.1. Preparation of the Photocatalysts**

The TiO<sub>2</sub> nanopowders were synthesized by a flame synthesis method using titanium tetra-isopropoxide (TTIP, Aldrich, 97 %) as a precursor. The TTIP was vaporized in an oil bath and delivered to the burner nozzle along with nitrogen gas as a carrier gas. The TTIP vapor was mixed with oxygen gas (an oxidizer) and methane gas (a fuel) in a burner and

combusted. The resultant TiO<sub>2</sub> nanoparticles loaded in the flowing product gas were transported to the collection chamber and separated from the product gas by filtration through a High Airflow Particulate Air (HAPA) filter. The obtained TiO<sub>2</sub> nanopowder (hereafter referred to as the as-synthesized TiO<sub>2</sub> nanopowder) was heat treated at 400, 500, 600, 700, 800 or 900°C for 1 h in air. Hereafter, the heat-treated powders are referred to as HTxxx, where xxx indicates the treatment temperature in Celsius degrees.

### **2.2.2. Characterization of crystalline and electronic structures**

The as-synthesized and heat-treated TiO<sub>2</sub> nanopowders were characterized using X-ray diffraction to identify the constituent phase(s) and particle size (XRD; Bruker D8 Advance), [17] and transmission electron microscopy to determine the shape and size of the particles (TEM; FEI Technai G<sup>2</sup>.)

The photo-excited structures of the TiO<sub>2</sub> nanopowders under UV light irradiation were directly observed by NEXAFS (Near-Edge X-ray Absorption Fine Structure; 7B1 KIST B/L at the Pohang Accelerating Laboratory (PAL), Korea). The powder samples were compacted into thin disks without using any polymer binder in order to prevent surface changes and additional effects associated with polymer binders. Two kinds of 8W UV-lamps, radiating UVA ( $\lambda = 320 - 400$  nm) and UVB ( $\lambda = 280 - 320$  nm), were installed

together in the NEXAFS chamber. The compacted disk was mounted on the specimen holder and positioned vertically beneath the UV lamp. The UV lamp was located at a distance of 50 cm from the disc, forming an angle of 15 degrees from the normal line. Prior to data acquisition, the NEXAFS chamber was evacuated to  $10^{-9}$  Torr. The incident beam was irradiated perpendicular to the substrate surface and the resultant photo-currents were recorded. The oxygen K-edge spectra were recorded for every sample under the same normal NEXAFS conditions prior to turning on the UV lamp. Additional oxygen K-edge spectra were obtained twice for every sample under sequential irradiation with UVA followed by UVB and these NEXAFS spectra are referred to as UVA and UVB. Just after turning off the UVB lamp, two additional scans of the compacted disk (called OFF1 and OFF2) were performed.

### **2.2.3. Photocatalytic performance**

The photoreactivity of the TiO<sub>2</sub> nanopowders was analyzed for the photo-degradation of 2-propanol into CO<sub>2</sub> on the TiO<sub>2</sub> photocatalysts under UV light irradiation. An amount of 50 mg of the TiO<sub>2</sub> nanopowder was suspended in a quartz cell containing an aqueous solution of 2-propanol ( $2.6 \times 10^{-3}$  mol·dm<sup>-3</sup>, 25 mL). Prior to UV light irradiation, the suspension was stirred for 30 min under oxygen atmosphere in the dark. The suspension

was then continuously stirred under oxygen atmosphere at 295 K and simultaneously irradiated with UV light ( $\lambda > 254$  nm) emitted from a 100 W high-pressure Hg lamp. An amount of 2 mL of the suspension was taken at regular intervals, filtered through a Millipore filter to separate the TiO<sub>2</sub> particles from the solution, and then analyzed by gas chromatography.

## **2.3 Results and discussion**

### **2.3.1. Characterization of the TiO<sub>2</sub> photocatalysts**

The morphology of the TiO<sub>2</sub> was investigated by TEM spectroscopy and TEM images of the as-synthesized and heat-treated TiO<sub>2</sub> nanopowders are shown in Fig. 2.1. The shape of the as-synthesized TiO<sub>2</sub> nanopowder is spherical, while that of HT500 changes to euohedral (Fig. 2.1b), while that of HT600, HT700, HT800 and HT900 change to an octagonal structure and/or cuboidal, as shown in Fig. 2.1c-f. The inset of each image shows the detailed morphologies of the TiO<sub>2</sub> particles in high magnification. It is clearly shown that inter-particle boundaries are evident on the TiO<sub>2</sub> after heat treatment above 700 °C. Thus, heat treatment of the TiO<sub>2</sub> nanopowders was observed to lead to changes in the morphology below 700 °C, probably due to the surface relaxation enhanced by surface



diffusion with the assisted mobility of the atoms.

Figure 2.2 shows the XRD patterns of the TiO<sub>2</sub> nanopowders as a function of the heat treatment temperatures. It is observed that the XRD patterns apparently change from an anatase to rutile phase at a heat treatment temperature of 800 °C. The particle size of the anatase and rutile phase of TiO<sub>2</sub>, which was calculated by the Scherer equation and the content of the rutile phase in each TiO<sub>2</sub> nanopowder are shown in Fig. 2.3. The as-synthesized TiO<sub>2</sub> involves ca. 98 % anatase phase together with ca. 2 % rutile phase. The transition state from anatase to rutile takes place at above 700 °C and is completed at 900 °C. The ratio of the rutile/anatase phase change was ca. 5 % at 700 °C, ca. 23 % at 800 °C, and ca. 99 % at 900 °C. On the other hand, the particle size of the anatase phase slightly increases from 50 nm to 55 nm by heat treatment up to 800 °C, while that of the rutile phase is estimated at ca. 67.5 nm at 900 °C.

In order to identify the presence of the anatase-rutile phase boundary in the HT800 powder, several dumbbell-type particles were closely observed by TEM analysis. Figure 2.4 shows the image of the HT800 powder dispersed in ethanol with ultrasonic treatment. A grain boundary composed of the anatase-rutile phase on the HT800 which chemically interact with each other is clearly observed.

### **2.3.2. Evaluation of the photocatalytic activity**

Figure 2.5 shows the photocatalytic reactivity for the degradation of 2-propanol on the as-synthesized and heat-treated TiO<sub>2</sub> nanopowders. It is observed that the degradation of 2-propanol takes place in proportion to the irradiation time on the TiO<sub>2</sub> photocatalysts under UV-light irradiation. The degradation rate of 2-propanol increases with an increase in the heat-treatment temperature up to 800°C, as shown in Fig. 2.5. Furthermore, the degradation rate of 2-propanol drastically decreased on HT900 due to the rutile phase, showing poor photocatalytic activity [17].

### **2.3.3. Photo-excited state of TiO<sub>2</sub> under UV-light irradiation**

NEXAFS was applied to investigate the photo-excited structure of TiO<sub>2</sub>. The electronic structures are affected in its photo-excitation energy by the wavelength of the incident light and such influences are assumed to change the NEXAFS spectra [18-20]. Figure 2.6 shows the NEXAFS spectra of the oxygen K-edge of the TiO<sub>2</sub> nanopowders. As can be seen, peaks at ca. 531-533 eV and 529-530.5 eV can be observed and they are attributed to the transitional absorption of the t<sub>2g</sub> and e<sub>g</sub> states of oxygen on TiO<sub>2</sub> having an anatase structure, respectively. In the present study, we determined the variations in the relative peak intensity between the t<sub>2g</sub> and e<sub>g</sub> levels as well as the peak position as a function of the

heat-treatment temperature. It should be noted that spectral quality during photo-irradiation showed serrated curves, irrespective of the samples. The as-synthesized TiO<sub>2</sub> nanopowder shows almost no change in the relative peak intensity of  $t_{2g}$  and  $e_g$  under different conditions (Fig. 2.6a). On the other hand, it was observed that UV-light irradiation of T600, T700 and T800 induces an increase in the peak of  $t_{2g}$ , whose height is close to that of  $e_g$ . Subsequently, T700 and T800 retained relative peak intensity at Stage OFF1 (over 30 min.). These results suggest that the recombination of the electrons with holes in the valance band is significantly suppressed, as shown in Fig 2.7. Furthermore, the phenomenon observed in HT900 is similar to that of the as-synthesized TiO<sub>2</sub> sample (data not shown).

#### **2.3.4. Relationship between photo-excited structures of TiO<sub>2</sub> and their photocatalytic performance**

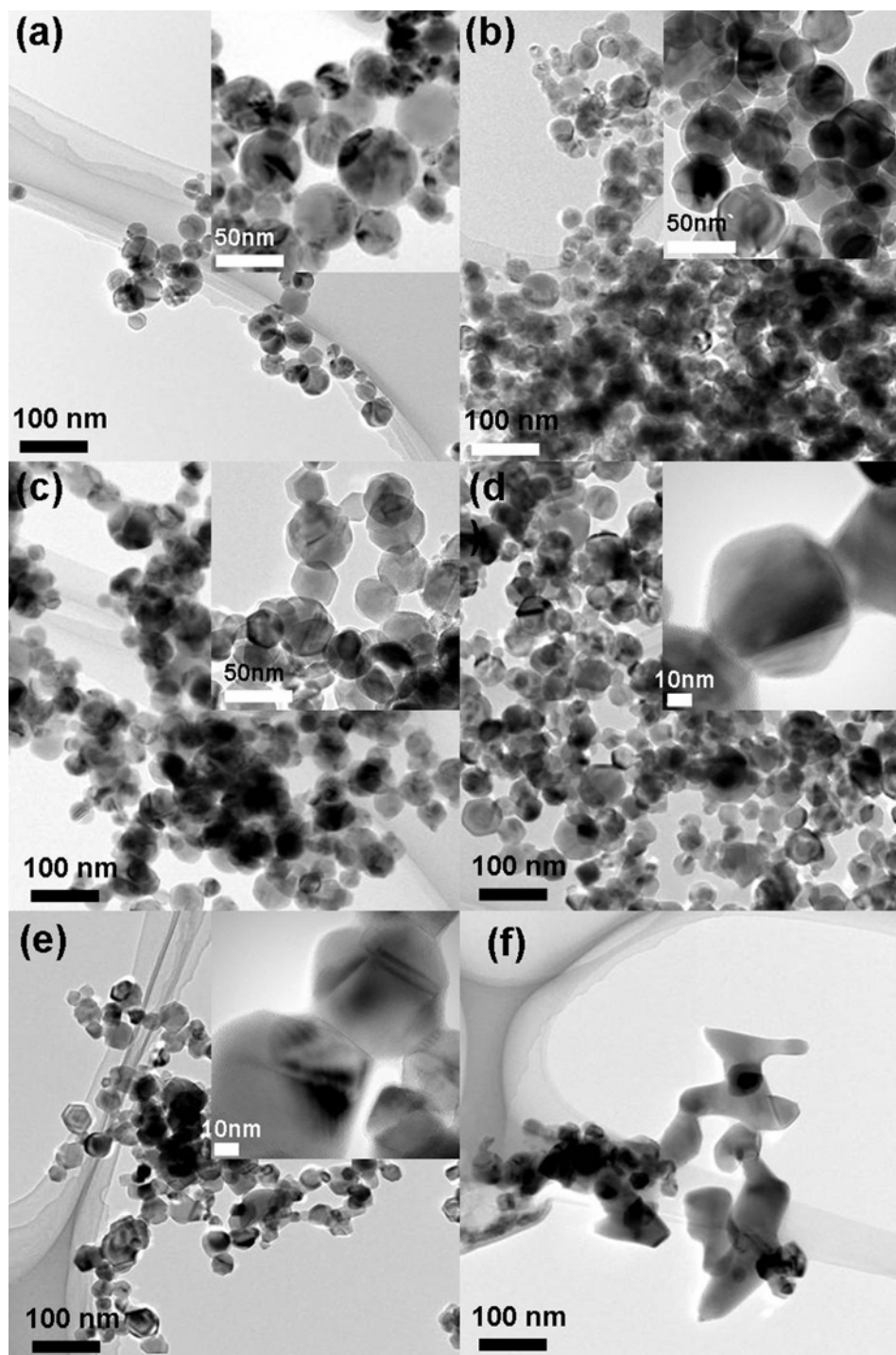
The rate of photocatalytic degradation increases in temperatures up to 700°C despite no significant differences in phase transition among HT500, HT600 and HT700, as shown in Fig. 2.2. From the TEM images, it is observed that an improvement in the photocatalytic degradation on the TiO<sub>2</sub> heat-treated up to 700°C is due to the changes in the particle shape of the TiO<sub>2</sub>. Further improvement in the photocatalytic properties of HT800 is

attributed to the presence of the anatase/rutile phase boundary [15-16], as shown in Figs. 2.2 and 2.3, since the boundary between the anatase and rutile phases at 800°C causes effective trapping of the photo-induced electrons, with a long lifetime in their excited states. On the other hand, drastic deactivation on HT900 is attributed to the dominant rutile phase.

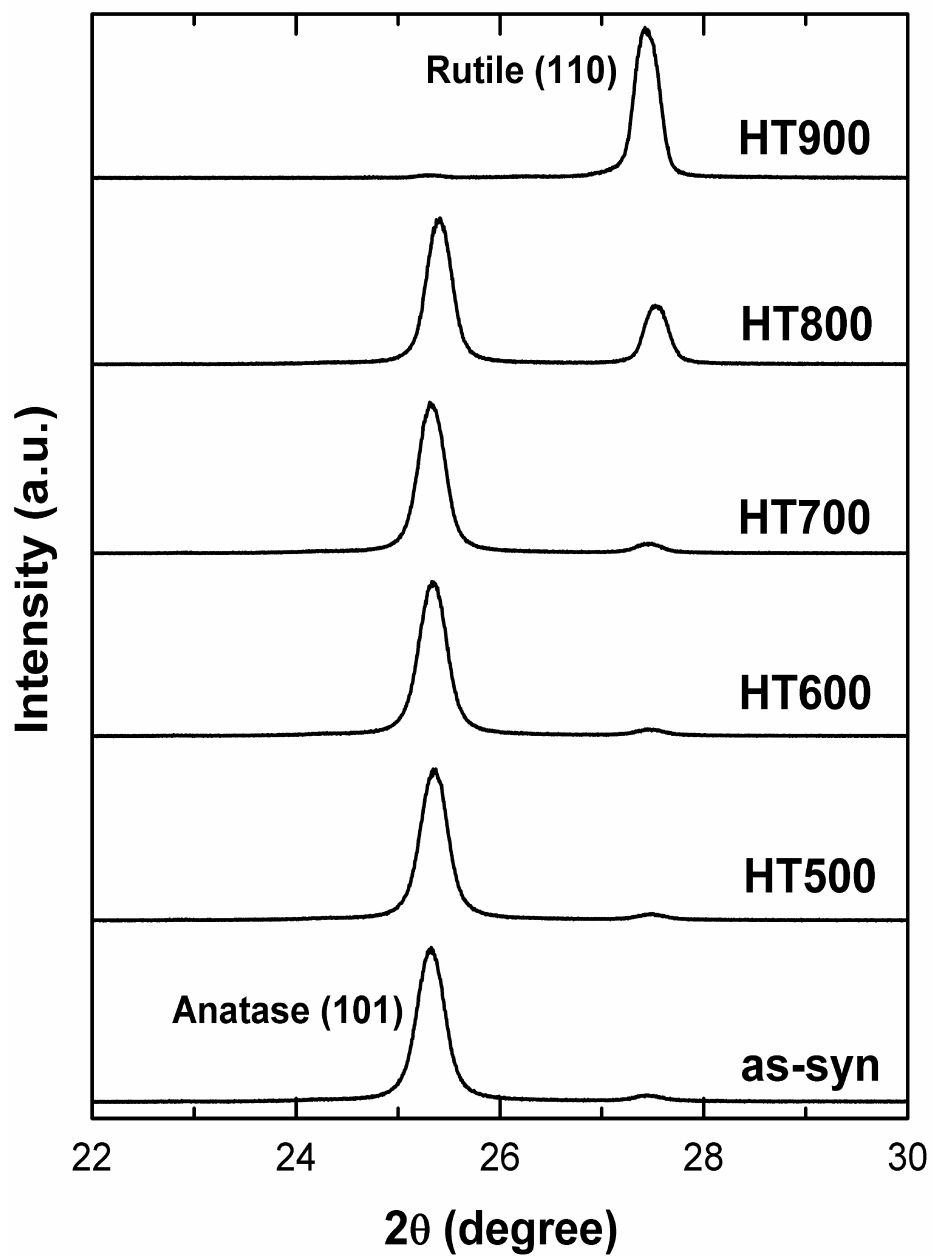
## **2.4. Conclusions**

Anatase phase-rich TiO<sub>2</sub> nanopowders containing small amounts of the rutile phase were fabricated by the flame method. The photoexcited states of the TiO<sub>2</sub> nanopowders were directly determined by in-situ NEXAFS measurements under UV light irradiation.

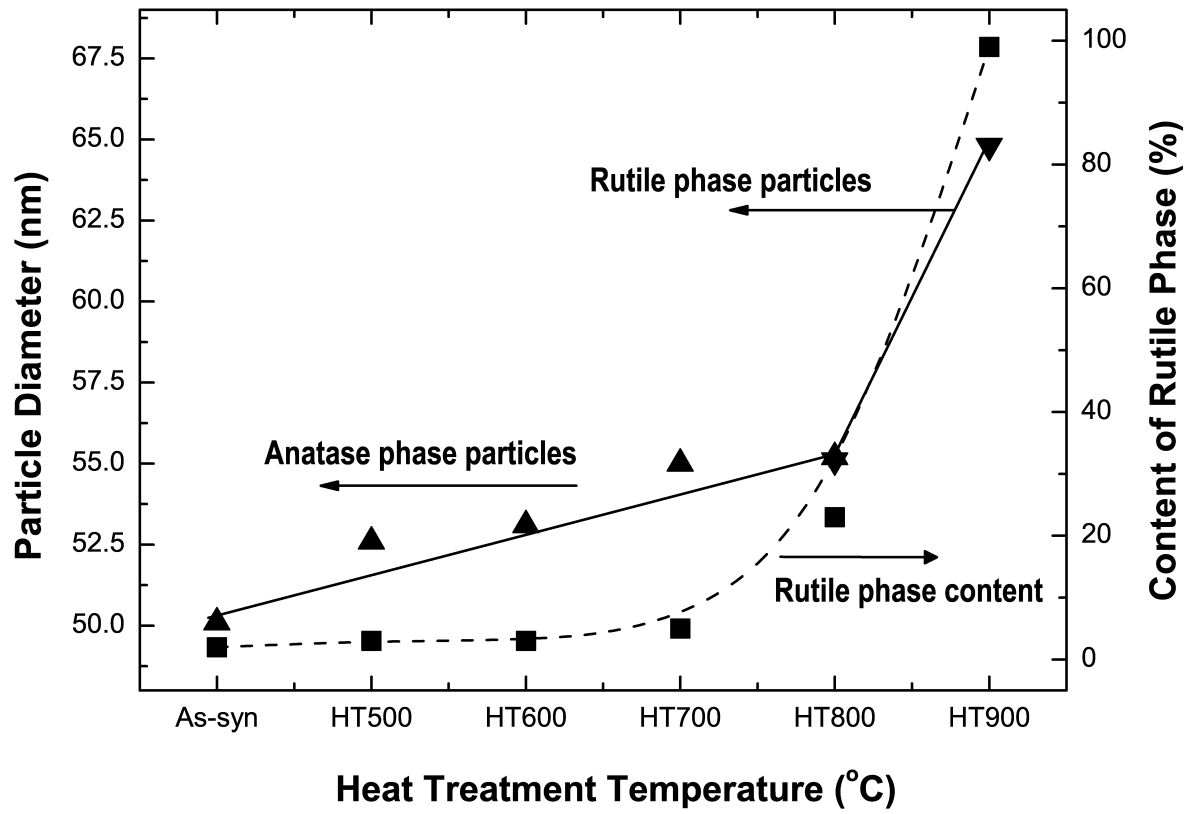
The present findings provide unprecedented direct experimental evidence by in-situ NEXAFS analysis showing that the electrons trapped in the anatase/rutile grain boundary suppress the recombination of electrons and holes, and in turn, this suppression of recombination directly contributes to an improvement in the photocatalytic reactivity for the decomposition of 2-propanol.



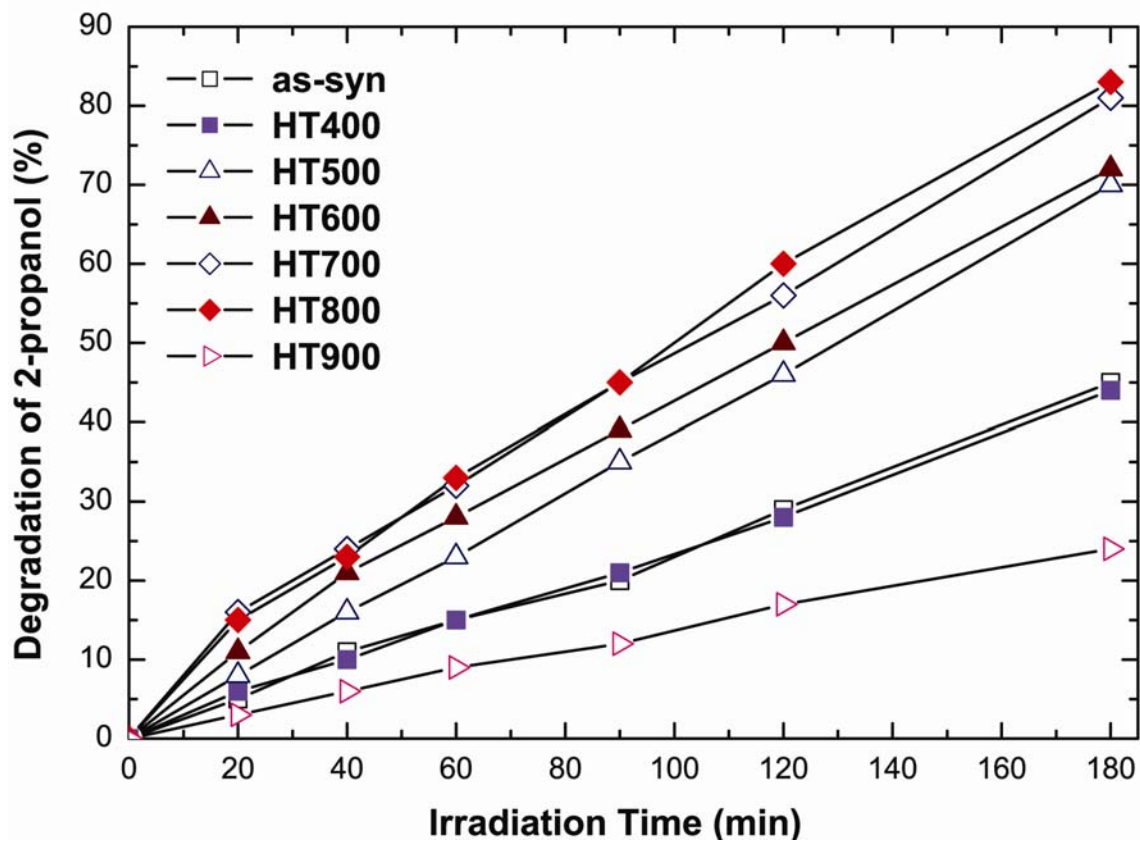
**Figure 2.1.** TEM images of TiO<sub>2</sub> nanopowders: (a) as-synthesized, (b) HT500, (c) HT600, (d) HT700, (e) HT800, (f) HT900.



**Figure 2.2.** XRD patterns of the as-synthesized and annealed TiO<sub>2</sub> nanopowders.

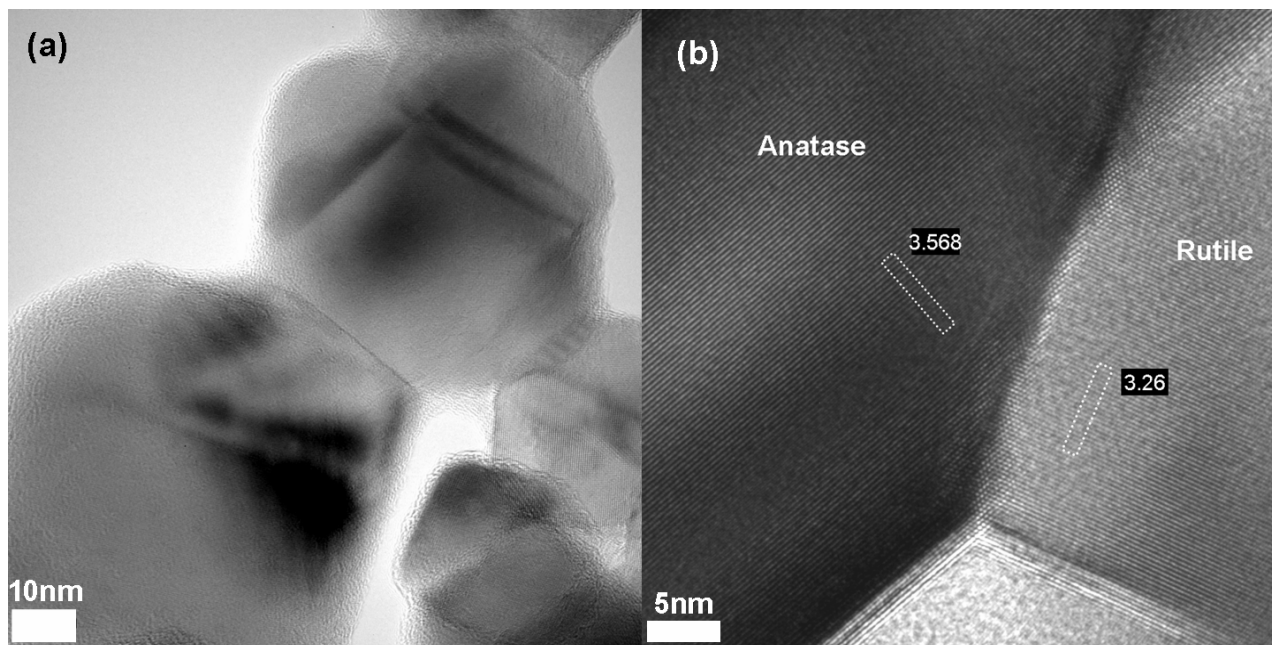


**Figure 2.3.** Variations in  $\text{TiO}_2$  particle size and content in the rutile phase for powders with heat treatment.

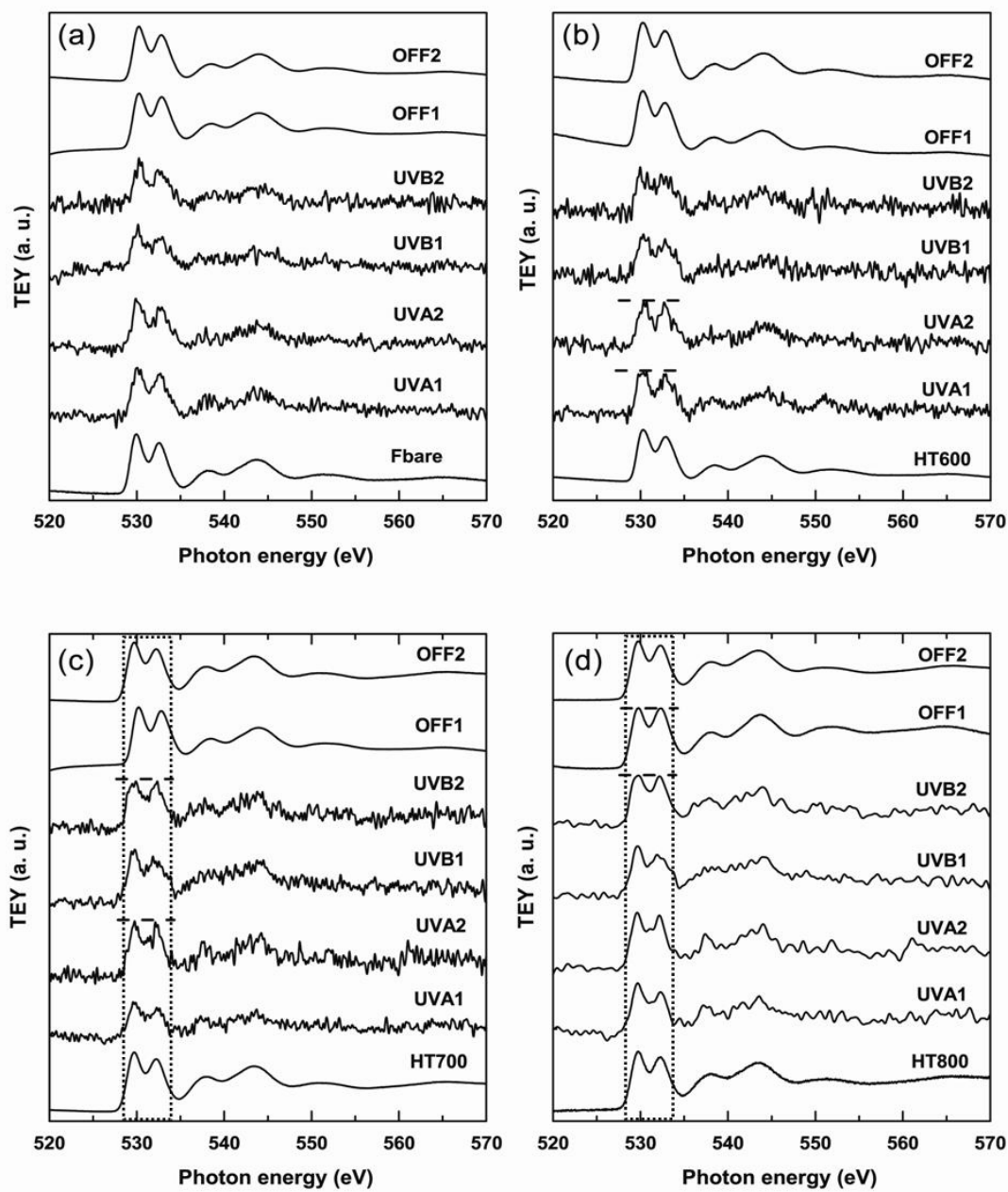


**Figure 2.4.** Photocatalytic degradation of 2-propanol under UV irradiation by various kinds of TiO<sub>2</sub> nanopowders (as-synthesized and heat treated).

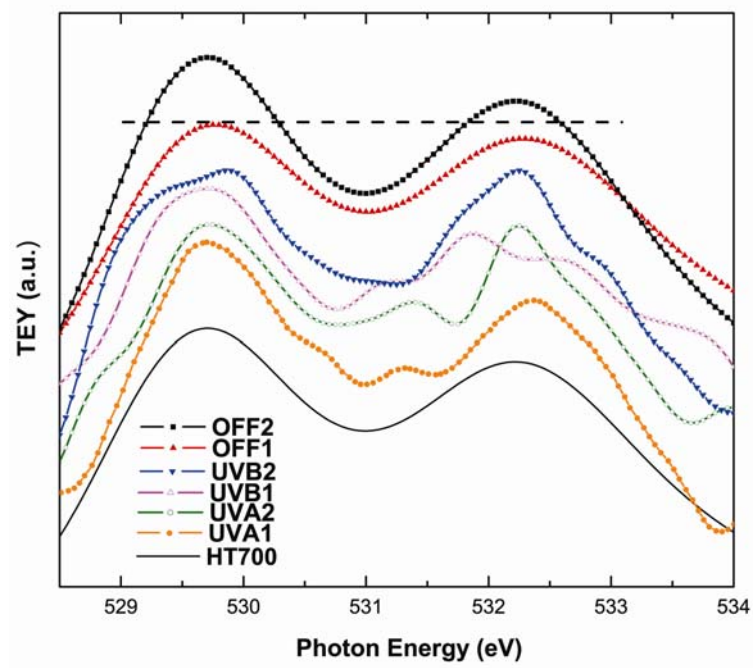




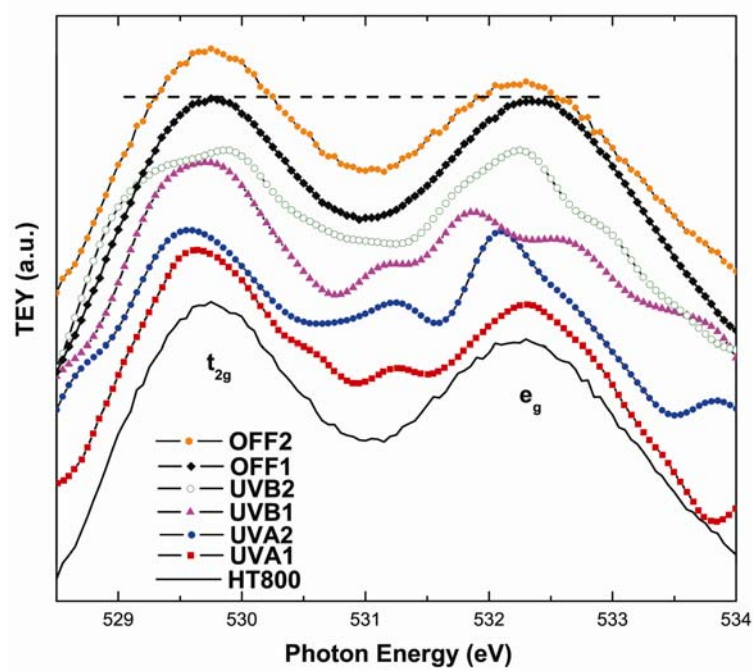
**Figure 2.5.** TEM images showing the anatase-rutile phase boundary in the TiO<sub>2</sub> nanopowder treated at 800 °C : (a) Low magnification, and (b) High magnification images.



**Figure 2.6.** Variations in oxygen K-edge spectra of  $\text{TiO}_2$  nanopowders under UV irradiation: (a) as-synthesized, (b) HT600, (c) HT700, (d) HT800.



(a)



(b)

**Figure 2.7.** NEXAFS O K-edge spectra of: (a) HT700, (b) HT800 (part of Figs. 6(c) and

(d))

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## **Chapter 3**

**Preparation of TiO<sub>2</sub> nano-particle photocatalysts  
by a multi-gelation method: The effect of pH change**

### **3. Preparation of TiO<sub>2</sub> nano-particle photocatalysts by a multi-gelation**

#### **method: The effect of pH change**

##### **3.1. Introduction**

TiO<sub>2</sub> catalysts can be applied for various purposes, e.g., as photocatalysts for pollution abatement [1-3], use in pigments [4], water splitting reactions [5], and solar cells [6], etc. They have been used extensively in environmental remediation processes due to their potential in oxidizing toxic organic compounds into CO<sub>2</sub> and water. However, continuous efforts to improve their reactivity are essential in order to realize their large scale global application and, presently, various methods are being investigated for the development of such highly efficient photocatalysts. Thus far, various preparation methods in which the photocatalytic activity of the TiO<sub>2</sub> particles depend on the preparation conditions have been reported [7-16]. In line with such work, we have prepared TiO<sub>2</sub> photocatalysts by employing various approaches such as the pH swing method [5, 11-16]. The pH swing method enables control of the intrinsic as well as extrinsic properties of the TiO<sub>2</sub> photocatalysts by a simple change in the pH of the reaction mixture during preparation [14]. An earlier pH swing method followed the principle of alternating the addition of TiCl<sub>4</sub> (as an acid solution) and aqueous ammonia (as a basic solution) to water at regular

intervals without adjusting the pH of the reaction mixture [14]. In the present study, the pH of the solution was kept constant at around 2 using HCl acid and at 8 using aqueous ammonia during preparation. The effect of the change in pH by a controlled pH swing method on the morphology of the TiO<sub>2</sub> particles as well as its application in the photocatalytic degradation ability of 2-propanol were investigated and compared with the photocatalytic properties of TiO<sub>2</sub> catalysts prepared by an uncontrolled pH swing method.

## **3.2. Experimental**

### **Catalyst preparation**

A TiCl<sub>4</sub> solution was prepared by mixing equal weights of TiCl<sub>4</sub> (obtained from Wako Chemicals, Japan) with crushed ice made from distilled water. TiO<sub>2</sub> catalysts were prepared by continuous heating and stirring of the TiCl<sub>4</sub> solution (500 mL) with an aqueous ammonia solution (14 wt%, 710 mL) under different pH swings at 353 K. A white precipitate of TiO<sub>2</sub> was prepared, filtered and dried at 393 K for 15 hrs. The dried TiO<sub>2</sub> was calcined at various temperatures with an electric furnace under a flow of air. Similarly, TiO<sub>2</sub> photocatalysts were prepared by a controlled pH swing method. In this method, 1 M HCl acid was used to bring down the pH to around 2 at each swing time and aqueous ammonia was used to adjust the pH to around 8.



## **Catalyst characterizations**

The diffuse reflectance absorption spectra of the photocatalysts were recorded with a Shimadzu UV-2200A spectrophotometer at 297 K. X-ray diffraction patterns of the photocatalysts were obtained with a Rigaku RDA-VA X-ray diffractometer using Cu K $\alpha$  radiation with a Nickel filter. The N<sub>2</sub> BET surface area of the TiO<sub>2</sub> catalysts was also determined. The pore volume and pore diameter were determined by a BET analyzer (Micromeritics, ASAP 2020, USA)..

## **Photocatalytic activity measurements**

The photocatalytic activity was investigated by comparing the reaction rates for the oxidative degradation of 2-propanol in which 2-propanol was seen to be completely oxidized into CO<sub>2</sub> and water on the TiO<sub>2</sub> photocatalysts under UV light irradiation in the presence of water and oxygen. The photocatalyst (50 mg) was suspended in a quartz cell with an aqueous solution of 2-propanol ( $2.6 \times 10^{-3}$  mol dm<sup>-3</sup>, 25 mL). Prior to UV light irradiation, the suspension was stirred for 30 min under oxygen atmosphere in dark conditions. The sample was then irradiated at 297 K using UV light ( $\lambda > 250$  nm) from a 100 W high-pressure Hg lamp with continuous stirring under oxygen atmosphere in the

system. At periodic intervals, 2 mL aliquots were taken from the system, centrifuged, and then filtered through a Millipore filter to remove the TiO<sub>2</sub> particles. The products were then analyzed by gas chromatography.

### **3.3. Results and discussion**

The XRD patterns of the TiO<sub>2</sub> photocatalysts showed the existence of well crystalline particles prepared by both uncontrolled pH swing (hereafter denoted as the flexible pH method) and controlled pH swing (hereafter denoted as the fixed pH method) and calcined above 450 °C (Fig. not shown). Figure 3.1 shows the anatase phase content of the TiO<sub>2</sub> particles prepared by both methods and calcined at 650°C. The catalysts calcined at 650°C under both preparation methods were comparable in terms of their catalytic properties. Thus, samples calcined at 650°C were used as the representative catalysts for a comparative study of both methods. Figure 3.1 clearly shows that the photocatalysts prepared by the fixed pH method (pH Fix) is able to prevent a phase transition, i.e., from anatase to rutile, irrespective of an increase in the number of pH swings and with calcination treatment at 650°C, whereas, in the flexible pH method (pH Fle), the anatase phase gradually increased with an increase in the number of pH swings [14]. For example, at 5 times pH swing, a 3 % anatase phase was formed for the pH Fle method calcined at

650°C, whereas a 75 % anatase phase was obtained for 20 times pH swings at the same calcination temperature. However, an 85 % anatase phase was observed in TiO<sub>2</sub> photocatalysts prepared by the pH Fix method at 800°C calcination up to 10 times pH swings. Thus, the pH Fix method could retain the anatase phase of the TiO<sub>2</sub> catalysts, irrespective of an increase in the number of pH swings and the calcination temperature, a significant observation of this study. The formation of the rutile phase for TiO<sub>2</sub> was observed only after 750°C calcination for pH Fix, as shown in Table 3.1. Photocatalysts prepared by the pH Fle method was reported to retain more of the anatase phase up to 600°C and 30 times pH swing [14], whereupon the anatase phase of TiO<sub>2</sub> changed to the rutile phase. When the catalysts were subjected to calcination above 600°C with the pH Fix method, a more anatase phase became evident up to a temperature of 750°C with up to 15 times pH swing (Table 3.1).

The particle size of the TiO<sub>2</sub> photocatalysts prepared by both methods increased with an increase in the number of pH swings (Fig. 3.2 and Table 3.1). This is due to the alternate addition of acid TiCl<sub>4</sub> and base aqueous ammonia during preparation of the TiO<sub>2</sub> catalysts at each swing time, in which small particles were dissolved by the acid solution and only large particles with high surface areas were retained. However, the average particle size of the catalysts prepared by pH Fix was found to be less than that by pH Fle,

as shown in Fig. 3.2. This is due to the dissolution of not only the smaller particles but also the large particles of  $\text{TiO}_2$  by the high concentration of the HCl acid, resulting in the formation of only small  $\text{TiO}_2$  particles with pH Fix. For pH Fle, the pH of the reaction mixture gradually became neutral when the number of pH swings increased to around 15 times. The effect of acid and alkaline was not very pronounced with pH Fle after a certain amount of pH swings, i.e., after a neutral pH was attained, however, the particles grew steadily with an increase in the pH swing numbers. It is worth noting that the surface area of the  $\text{TiO}_2$  particles prepared by the pH Fle method increased with an increase in the pH swing numbers [Fig. 3.3], whereas, the reverse trend was observed for  $\text{TiO}_2$  prepared by the pH Fix method. Although the particle size gradually increased with an increase in pH swings for both methods, not much influence was observed on the surface area for the particles prepared by pH Fix. This may be due to the existence of a small particle pore size and pore volume as well as the formation of a rutile phase at higher calcination temperatures. The rutile particles are aggregated larger particles responsible for a decrease in the surface area at higher calcination temperatures (Table 3.1), while at the same time, the pore volume and pore diameter have a strong influence on the morphology of the  $\text{TiO}_2$  particles. Figures 3.4 and 3.5 clearly show that both the pore volume and pore diameter of the  $\text{TiO}_2$  particles increased tremendously with an increase in the pH swing numbers with

the pH Fle method, whereas, only a slight increase in the pore volume and diameter of the TiO<sub>2</sub> particles were observed for the pH Fix method. The high pore volume and pore diameter of the particles were, thus, seen to be responsible for the high surface area of the particles prepared by pH Fle and the smaller pore volume and pore diameter were attributed to the smaller surface area of TiO<sub>2</sub> particles prepared by pH Fix. Well-crystalline TiO<sub>2</sub> particles were formed when the number of pH swings increased from 5 to 30. This is also another reason for the decrease in the surface area of TiO<sub>2</sub> particles prepared by the pH Fix method (Fig. 3.3).

The results of 2-propanol oxidation were investigated for the photocatalytic degradation ability of the catalysts prepared by these two methods and the results are shown in Fig. 3.6. The photocatalytic activity of the TiO<sub>2</sub> catalysts prepared by 20 times pH swings and calcined at 650°C showed a higher rate for the degradation of 2-propanol in comparison with other catalysts prepared by pH Fle with different pH swing numbers and calcination temperatures. Similarly, the catalysts prepared by 25 and 30 pH swing times and calcined at 650°C showed high activity for the degradation of 2-propanol using the pH Fix method, although it was still found to be less than the catalysts prepared by the pH Fle method, as shown in Fig. 3.6. A combination of both anatase and rutile phases have been reported to enhance the reaction rate for the degradation of organic pollutants to a

certain extent [17-19]. In this study, the catalysts prepared by 15 pH swing times by pH Fix and calcined at 750°C possessed a mixture of anatase and rutile phases in a ratio similar to the P-25 catalyst (Table 3.1). However, the catalytic activity was found to be less than the catalysts prepared by 30 times pH swings and calcined at 650°C which consisted of 94 % anatase. With pH Fle, the catalysts calcined at 650°C possessed an anatase/rutile ratio of around 75/25 with high pore volume and pore diameter, showing a high efficiency for the degradation of 2-propanol (78 %) (Fig. 3.6 and Table 3.2). The TiO<sub>2</sub> catalysts possessing an anatase/rutile ratio of around 70/30, with less pore volume and pore diameter than the catalysts prepared by 20 times pH swings, did not show high activity for the degradation of 2-propanol (58 %) (Table 3.2). Moreover, less pore volume and pore diameter were observed for the catalysts prepared with 5 times pH swings than with 20 times pH swings (Table 3.2). An anatase/rutile ratio of around 70/30, thus, had no effect on the photocatalytic activity for the degradation of organic compounds. This is clearly shown in Table 3.2 in which TiO<sub>2</sub> particles with a high pore volume and pore diameter showed excellent activity for the degradation of 2-propanol. These results reveal not only that the anatase/rutile phase is an important parameter for the catalytic reactions but also that other important parameters such as pore volume and pore diameter are equally important for the photocatalytic degradation reactions. The pH Fle method enabled

the preparation of more efficient TiO<sub>2</sub> photocatalysts comparable to P-25 (Fig. 3.6) than the pH Fix method, especially for photocatalytic degradation reactions as well as control of the morphology of the particles. With pH Fix, the particle size increased at a calcination temperature of 750°C (Table 3.1), although the surface area did not increase proportionally and the pore size as well as pore volume were found to be less than the catalysts calcined at 650 and 700°C. These results indicate that in addition to the anatase/rutile phase ratio, the particle size and surface area of the particles, and the pore volume and pore diameter are major factors in realizing the efficient photocatalytic degradation of organic compounds.

### **3.4. Conclusions**

TiO<sub>2</sub> photocatalysts prepared by a controlled pH swing method could retain the anatase phase even at calcination temperatures of 750°C at high pH swing numbers. However, other important parameters such as particle size, surface area, pore volume, pore size as well as the anatase/rutile phase ratio could not be controlled well by this method. The addition of HCl acid during preparation showed detrimental effects on the morphology of the particles. On the other hand, TiO<sub>2</sub> catalysts prepared by an uncontrolled pH swing method showed better performance, especially in the control of such important parameters.

The results of 2-propanol oxidation showed that control of the anatase/rutile ratio, the pore volume as well as pore diameter of the TiO<sub>2</sub> nano-particles are important factors in realizing the efficient photocatalytic degradation of organic compounds.



**Table 3.1.**Physicochemical properties of TiO<sub>2</sub> photocatalysts prepared by the fixed pH method

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Calcination temperature (°C)	Number of pH swings	Surface area (m <sup>2</sup> /g)	Particle size (nm)	Anatase:rutile	Pore volume (cc/g)	Pore diameter (nm)
700	15	31	14	93:07	0.139	19
700	20	32	18	90:10	0.151	20
700	25	33	20	89:11	0.154	21
700	30	38	22	93:07	0.200	23
750	15	20	19	85:15	0.102	22
750	20	16	25	55:45	0.099	24
750	25	15	26	60:40	0.066	25
750	30	15	27	50:50	0.100	26

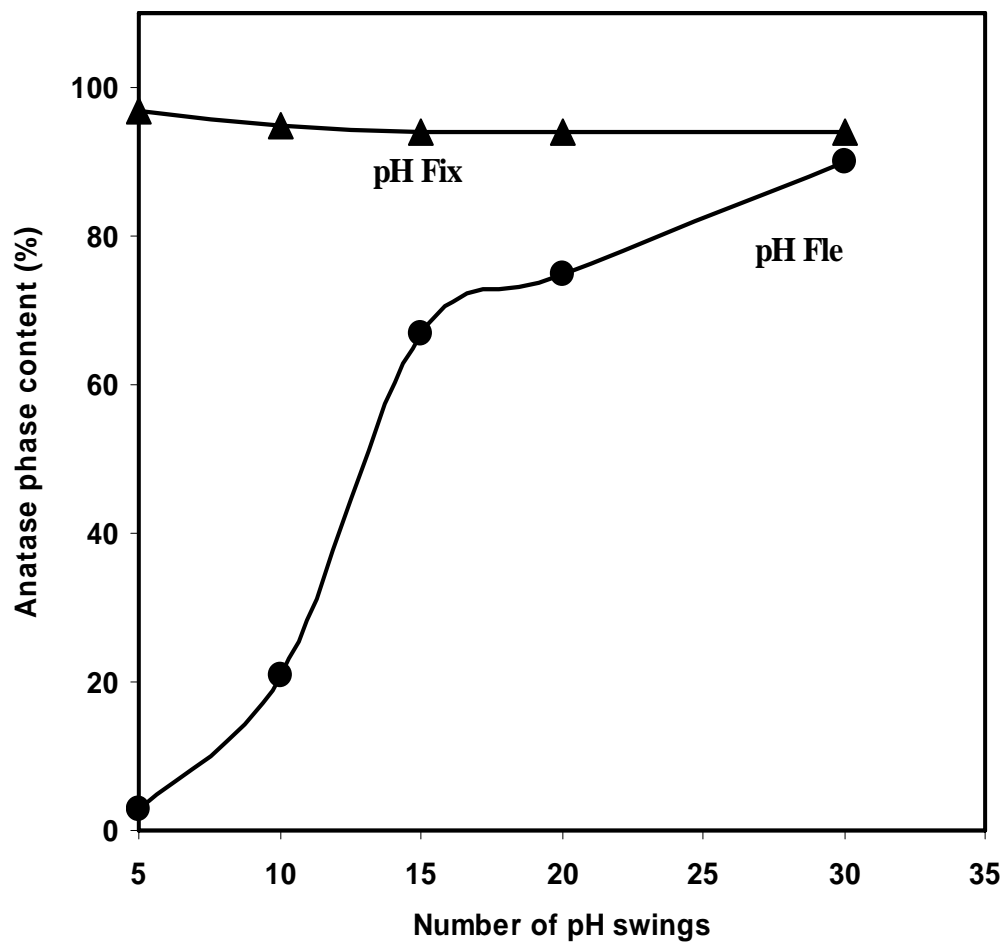
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**Table 3.2.**Physicochemical properties of TiO<sub>2</sub> photocatalysts prepared by the flexible pH method

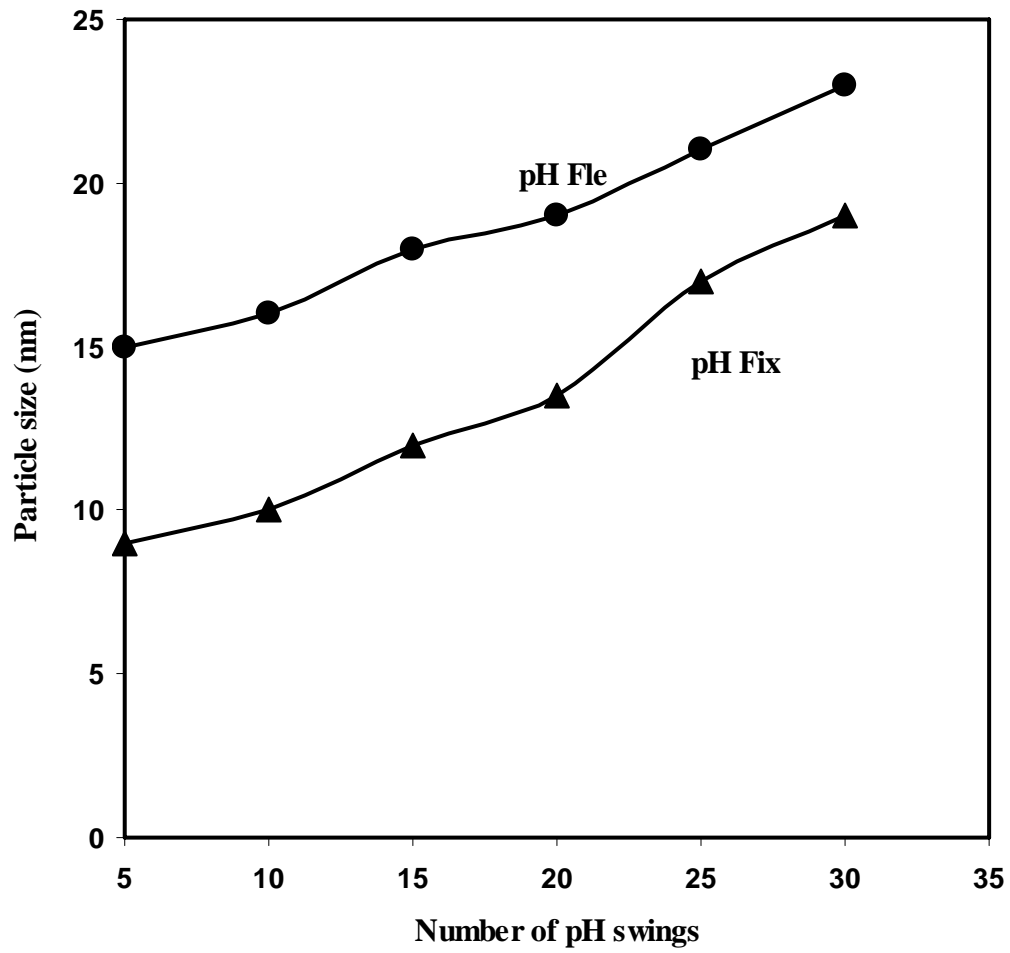
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Number of pH swings	Calcination temperature (°C)	Anatase:rutile	Pore-volume (cc/g)	Pore-diameter (nm)	2-propanol degradation (%)
5	550	70:30	0.135	13	58
10	600	60:40	0.293	31	64
15	650	67:33	0.515	57	77
20	650	75:25	0.513	64	78

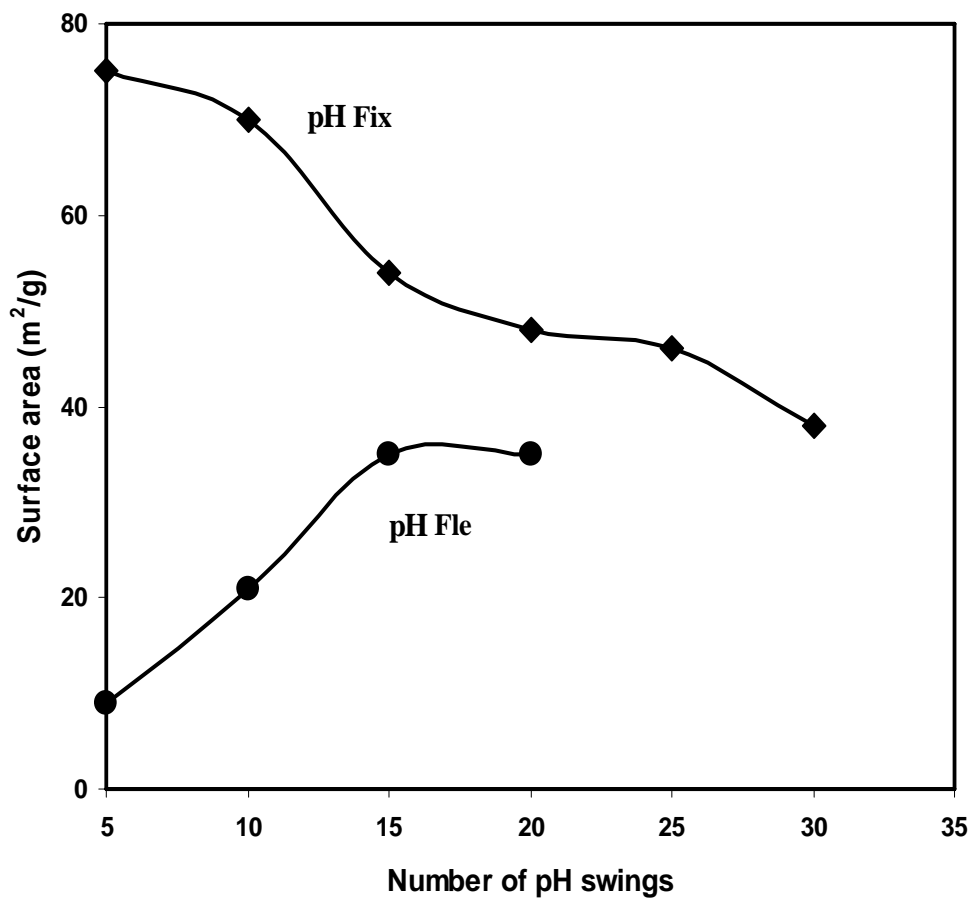
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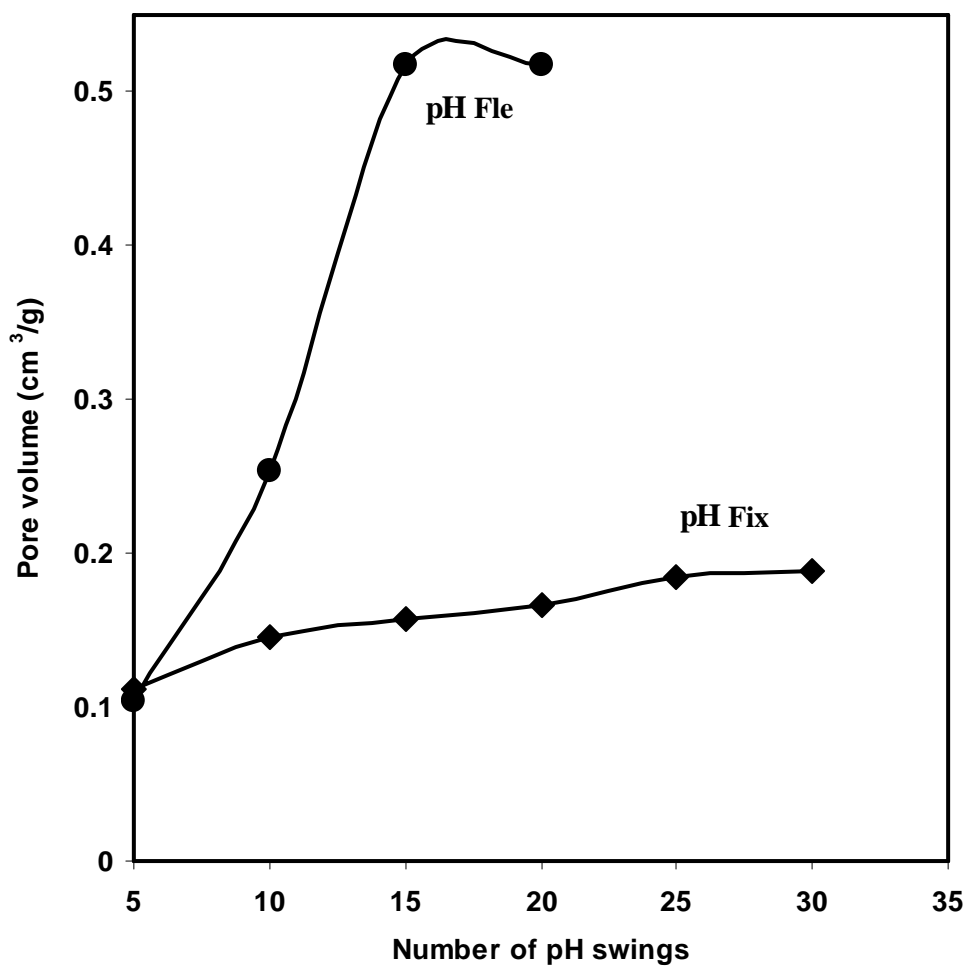
**Figure 3.1.** Effect of the number of pH swings on the anatase phase content of  $\text{TiO}_2$  particles calcined at  $650^\circ\text{C}$  when prepared by the flexible (pH Fle) and fixed pH swing (pH Fix) methods.



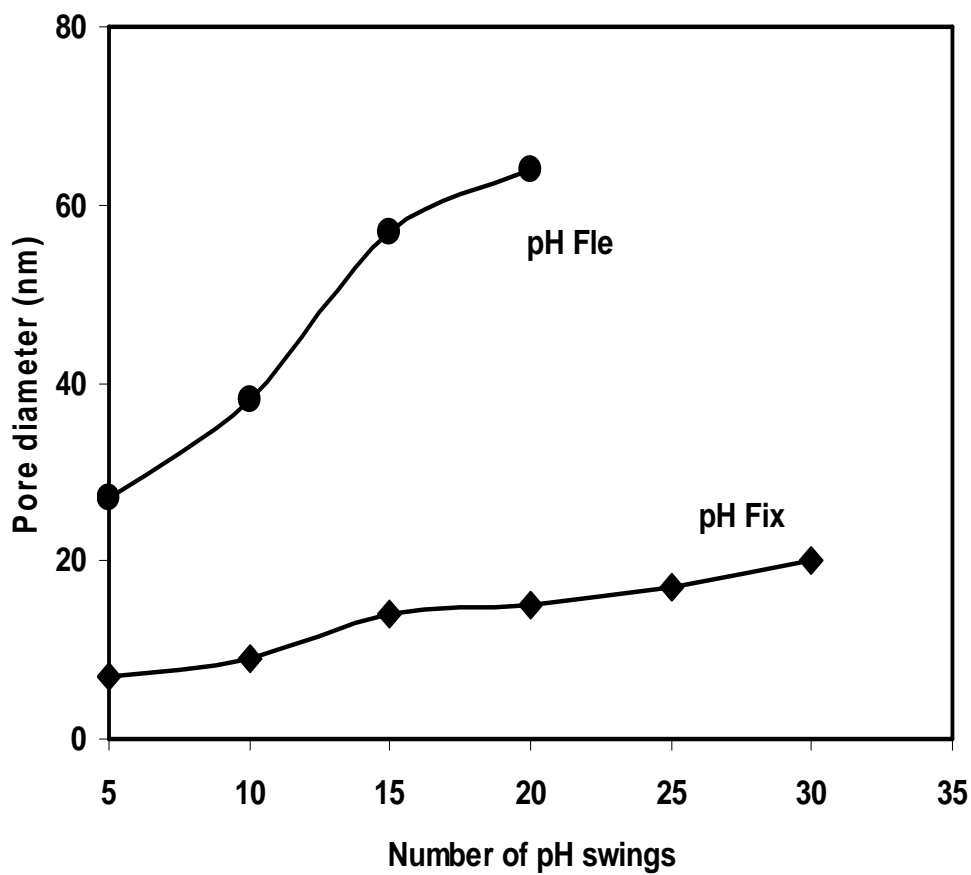
**Figure 3.2.** Particle size of photocatalysts calcined at 650°C versus the number of pH swings prepared by the flexible (pH Fle) and fixed pH swing (pH Fix) methods.



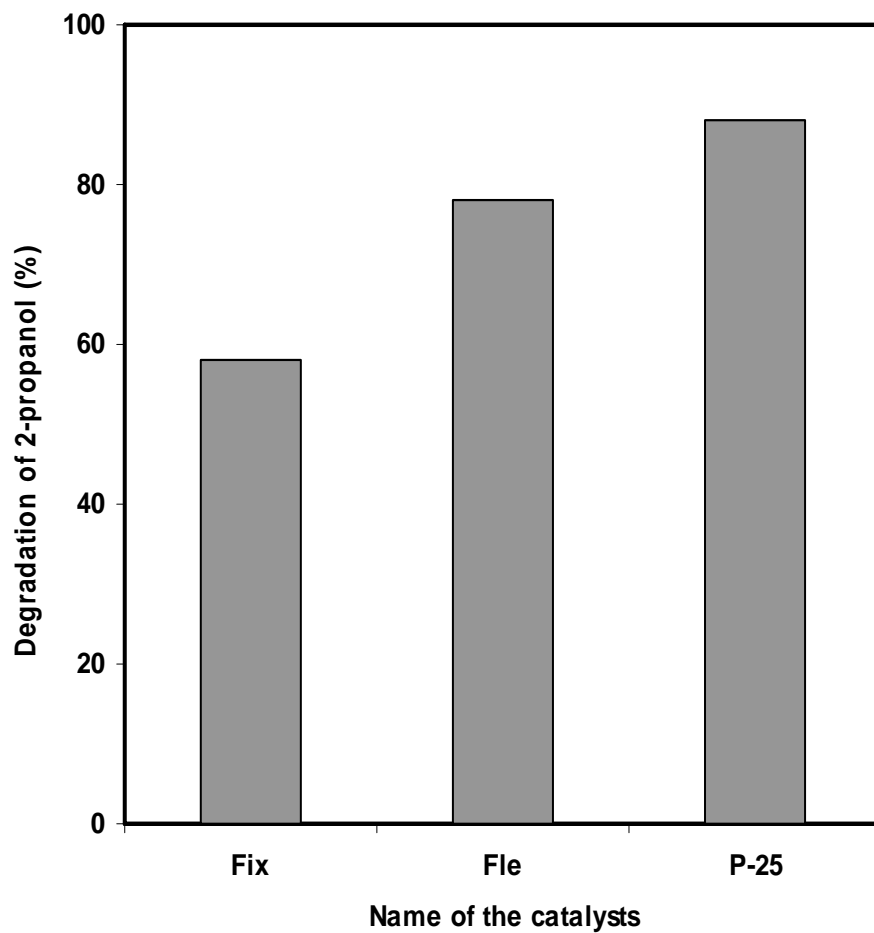
**Figure 3.3.** Specific surface area of photocatalysts calcined at 650°C versus the number of pH swings prepared by the flexible (pH Fle) and fixed pH swing (pH Fix) methods.



**Figure 3.4.** Pore-volume of photocatalysts calcined at 650°C versus the number of pH swings prepared by the flexible (pH Fle) and fixed pH swing (pH Fix) methods.



**Figure 3.5.** Pore-diameter of photocatalysts calcined at 650°C versus the number of pH swings prepared by the flexible (pH Fle) and fixed pH swing (pH Fix) methods.



**Figure 3.6.** Comparison of the oxidative degradation of 2-propanol into  $\text{CO}_2$  and  $\text{H}_2\text{O}$  (UV irradiation for 4 hrs at 297 K) using catalysts prepared by the flexible (pH Fle) & fixed pH swing (pH Fix) methods and P-25  $\text{TiO}_2$ .



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## **Chapter 4**

### **Enhancement of the Photocatalytic Reactivity of TiO<sub>2</sub> Nano-particles by Simple Mechanical Blending with Hydrophobic MOR Zeolites**

## **4. Enhancement of the Photocatalytic Reactivity of TiO<sub>2</sub> Nano-particles by Simple Mechanical Blending with Hydrophobic MOR Zeolites**

### **4.1. Introduction**

TiO<sub>2</sub> photocatalysts have been widely studied for the purification of air, water, and soil polluted with organic compounds for their potential to completely decompose harmful organic compounds into CO<sub>2</sub> and H<sub>2</sub>O under UV light irradiation.[1-4] Various air-cleaning and deodorization systems equipped with TiO<sub>2</sub> photocatalysts and UV light sources that reduce volatile organic compounds (VOCs) such as aldehydes, carboxylic acids and aromatic compounds which can cause the so-called “sick house syndrome” are now commercially available. However, the removal efficiencies of air-cleaning systems for odorant compounds in the home environment still needs improvement, preferably by as simple and low-cost a method as possible. Although the deposition of small amounts of Pt on TiO<sub>2</sub> catalyst surfaces is generally known to enhance the photocatalytic reactivity,[5-9] Pt compounds are too costly for common use in home electrical appliances. On the other hand, the hybridization of adsorbents such as zeolites [10-12] and activated carbon [13-15] with TiO<sub>2</sub> particles has been reported to show elevated photocatalytic reactivity, especially for the decomposition of gaseous organic compounds. In a previous report,[16] we have

shown that hybridized TiO<sub>2</sub>/MOR catalysts prepared by a simple impregnation method also exhibit high photocatalytic reactivity for the complete oxidation of gaseous acetaldehyde as compared to pure untreated TiO<sub>2</sub> catalysts since MOR zeolite powders are able to efficiently adsorb acetaldehyde molecules diffused in wide spaces and then smoothly supply them onto the TiO<sub>2</sub> photocatalyst surfaces.

In this work, TiO<sub>2</sub> nano-particles were mechanically blended with the hydrophobic MOR zeolite in a simple preparation method in order to maximize the photocatalytic performance of commercial TiO<sub>2</sub> powders as well as reduce the preparation cost. The photocatalytic reactivity of the blended TiO<sub>2</sub>/MOR systems were then evaluated for the complete oxidation of gaseous acetaldehyde with O<sub>2</sub> under UV light irradiation.

## 4.2. Experimental

TiO<sub>2</sub> nano-powdered photocatalysts (SSP-25, anatase phase, SSA<sub>BET</sub> = ca. 270 m<sup>2</sup>/g) and the highly siliceous H<sup>+</sup>-type MOR zeolite (HSZ-HOA890, SiO<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> = ca. 1880, SSA<sub>BET</sub> = ca. 370 m<sup>2</sup>/g) were purchased from Sakai Chemical Industry Co., Ltd. and Tosoh Co., Ltd., respectively. TiO<sub>2</sub>/MOR photocatalysts having different TiO<sub>2</sub> content were obtained by a simple mechanical blending of these two powder samples in an agate mortar for 0 - 60 min [referred to as TiO<sub>2</sub>/MOR(A)]. For comparison, a different type of

TiO<sub>2</sub>/MOR photocatalysts was also prepared by the ultrasonic aqueous suspension of these two powder samples [referred to as TiO<sub>2</sub>/MOR(B)]. The TiO<sub>2</sub> nano-powders blended with the MOR zeolites were then characterized by XRD (Shimadzu, XRD-6100) and diffuse reflectance UV-vis absorption (Shimadzu, UV-2200A) measurements at room temperature.

The photocatalytic reactivity of the blended TiO<sub>2</sub>/MOR samples was evaluated for the decomposition of gaseous acetaldehyde in the presence of O<sub>2</sub> under UV light irradiation. The TiO<sub>2</sub>/MOR catalysts (50 mg) were placed onto a flat bottom quartz cell (volume, ca. 33 cm<sup>3</sup>). The volume of the reaction area including the cell volume was ca. 100 cm<sup>3</sup>. Before photoreactions were carried out, the catalysts were degassed at 723 K for 2 h, treated in sufficient amounts of O<sub>2</sub> (ca. 6.7 kPa) at the same temperature for 2 h, and then degassed at 373 K for 2 h up to a 10<sup>-5</sup> kPa range. A gas mixture of CH<sub>3</sub>CHO (0.27 kPa), O<sub>2</sub> (1.07 kPa), and H<sub>2</sub>O (0 - 1.33 kPa) was then introduced into the reaction cell. The amount of acetaldehyde introduced into the reaction cell was calculated as ca. 8 μmol (ca. 1500 ppm). After an adsorption equilibrium was reached, UV light was irradiated at 275 K by a 100 W high-pressure Hg lamp (Toshiba, SHL-100UVQ-2) through a cutoff filter (Toshiba Glass, UV-27, λ>270 nm, ca. 1-2 mW/cm<sup>2</sup>). To avoid the heating effect from the UV lamp, the photocatalysts in the quartz cell were cooled in ice water during the photoreactions. The amount of CO<sub>2</sub> produced and acetaldehyde decomposed were analyzed by TCD and

FID by gas chromatography (Shimadzu, GC-14A).

### **4.3. Results and Discussion**

Figure 4.1 shows the XRD patterns of the TiO<sub>2</sub> nano-powders blended with MOR zeolites of different TiO<sub>2</sub> contents. The TiO<sub>2</sub> nano-powders showed typical diffraction patterns attributed to the (101) phase of an anatase structure at around 26 degrees. The primary particle size of the TiO<sub>2</sub> nano-powder (SSP-25) could be estimated at ca. 8 nm by the Scherrer's equation. On the other hand, all diffraction patterns for the MOR zeolite which could be assigned to a MFI structure much sharper than the TiO<sub>2</sub> nano-powder, showing largely grown zeolite crystals. When small amounts of TiO<sub>2</sub> nano-powders were mechanically blended with such largely grown zeolite particles even for only 5 min, it became difficult to observe the diffraction patterns attributed to the TiO<sub>2</sub> nano-powders. However, as the TiO<sub>2</sub> content increased up to 20 wt%, a broad diffraction peak attributed to the anatase (101) phase could be observed at around 26 degrees.

The diffuse reflectance UV-vis absorption spectra of the TiO<sub>2</sub> nano-particles mechanically blended with the siliceous MOR zeolite and non-porous pure SiO<sub>2</sub> powder are shown in Figs. 4.2-A and 4.2-B, respectively. The absorption edges of the TiO<sub>2</sub>

nano-powders blended with the MOR zeolites were observed at around 380 - 400 nm. When smaller amounts of TiO<sub>2</sub> powders than 5 wt% were blended with the siliceous MOR or SiO<sub>2</sub> powders, some portion of the incident light was found to pass through the powder samples of several millimeters thickness due to the high transparency of the MOR and SiO<sub>2</sub> powders. When the amount of TiO<sub>2</sub> nano-powders blended with the zeolite powders reached about 10 - 20 wt%, the incident light could not penetrate the mixed powder samples, suggesting efficient irradiation of UV light onto the entire TiO<sub>2</sub> nano-particles. However, since the absorption coefficient of the TiO<sub>2</sub> powder in UV light regions is known to be very high, as the fraction of the TiO<sub>2</sub> powders to the zeolite powders increased, the incident light could not be irradiated onto the backside of the TiO<sub>2</sub> particles. These results clearly indicate that an important role of the siliceous MOR zeolite or SiO<sub>2</sub> powders is the efficient irradiation of incident UV light onto all of the TiO<sub>2</sub> nano-particles without any loss of light intensity.

The oxidation of gaseous acetaldehyde on TiO<sub>2</sub> photocatalysts hardly proceeded in the absence of a H<sub>2</sub>O vapor, however, the photocatalytic reaction was dramatically enhanced by adding small amounts of H<sub>2</sub>O vapor [16]. The photocatalytic oxidation reactions of gaseous acetaldehyde with O<sub>2</sub> under UV light irradiation ( $\lambda > 270$  nm) over the TiO<sub>2</sub> nano-particles blended with the hydrophobic MOR zeolite (SiO<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> = 1880) were, thus,



evaluated in the presence of a H<sub>2</sub>O vapor. At first, the effect of the mechanical mixing times of these two different powder samples on the photocatalytic reactivity of TiO<sub>2</sub>/MOR(A) was evaluated from the viewpoint of the dispersion of the TiO<sub>2</sub> nano-particles within the zeolite powders. As shown in Fig. 4.3, although the increased dispersion of the TiO<sub>2</sub> nano-particles onto the powders after mixing up to 60 min could be confirmed by SEM observations (not shown), the photocatalytic reactivity could hardly be enhanced. For comparison, the mixed TiO<sub>2</sub>/MOR(B) powders were prepared from an ultrasonically aqueous suspension. However, as shown in Fig. 4.4, the photocatalytic reactivity of TiO<sub>2</sub>/MOR(B) was almost equivalent to the TiO<sub>2</sub>/MOR(A) prepared by a simple mechanical blending method. Moreover, it is notable that the TiO<sub>2</sub> and zeolite powders without mixing in an agate mortar showed slightly less photocatalytic reactivity as compared to the mechanically blended TiO<sub>2</sub>/MOR(A). These results clearly indicate that the photocatalytic reactivity of the TiO<sub>2</sub> nano-particles is easily improved by simple mechanical blending with a hydrophobic zeolite powder as an adsorbent material.

Figure 4.5 shows the effects of the TiO<sub>2</sub> content on the photocatalytic reactivity of the mechanically blended TiO<sub>2</sub>/MOR(A) for the complete oxidation of gaseous acetaldehyde with O<sub>2</sub> in the presence of H<sub>2</sub>O vapor under UV light ( $\lambda > 270$  nm) irradiation. TiO<sub>2</sub> nano-particles of ca. 5 - 15 wt% mechanically blended with the siliceous MOR zeolite of

hydrophobic character showed almost twice as high photocatalytic reactivity as compared to the pure untreated TiO<sub>2</sub> nano-particles. Also, as reported in a previous work, the siliceous MOR zeolite can work as a good adsorbent to concentrate gaseous acetaldehyde in hydrophobic cavities.[16] Since the siliceous zeolite does not have Brønsted acid sites, which work as strong adsorption sites for polar molecules such as H<sub>2</sub>O molecules,[17] the acetaldehyde molecules concentrated within the zeolite cavities could smoothly diffuse on the catalyst surfaces. In addition, as mentioned from the results of UV-vis absorption measurements, the incident UV light was efficiently irradiated on the entire TiO<sub>2</sub> photocatalyst of ca. 5 - 15 wt% blended with the MOR zeolite due to the high transparency of the zeolite powders in UV-vis light regions. In order to verify the role of the zeolite powders, the photocatalytic reactivity of the TiO<sub>2</sub> nano-particles mechanically blended with non-porous silica powders were also investigated. Although the non-porous silica (SSA<sub>BET</sub> = less than 10 m<sup>2</sup>/g) adsorbed only small amounts of gaseous acetaldehyde molecules as compared to the MOR zeolite, transparent silica powders can work as an efficient diluting material for the TiO<sub>2</sub> nano-particles. In this case, the non-porous silica powders did not show any condensation effect for the acetaldehyde molecules. Small amounts of the TiO<sub>2</sub> nano-particles blended with the non-porous silica powders, thus, showed low photocatalytic reactivity. From these results, the critical role of the

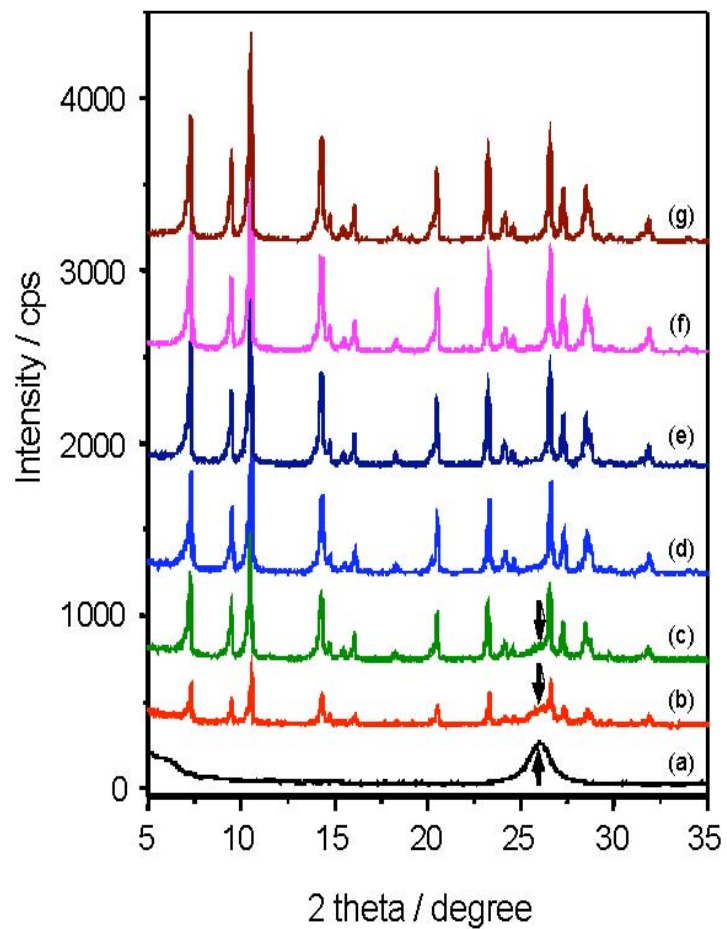
hydrophobic MOR zeolite to enhance the photocatalytic reactivity of TiO<sub>2</sub> nano-particles can be concluded to be: (i) the condensation effect for gaseous acetaldehyde molecules near the TiO<sub>2</sub> photocatalytic sites; and (ii) the appropriate diluent effect of the TiO<sub>2</sub> photocatalysts as an intense absorber of UV light with highly transparent zeolite powders.

For further verification of the condensation of gaseous acetaldehyde molecules, the photocatalytic oxidation reaction of different initial concentrations of acetaldehyde over the TiO<sub>2</sub> nano-particles and TiO<sub>2</sub>/MOR(A) were compared. As shown in Fig. 4.6, when only TiO<sub>2</sub> nano-particles were applied for the photocatalytic oxidation of gaseous acetaldehyde, the total conversion of acetaldehyde into CO<sub>2</sub> and H<sub>2</sub>O was found to decrease with a decrease in the initial pressure of acetaldehyde. On the other hand, TiO<sub>2</sub> nano-particles mechanically blended with the hydrophobic zeolite powders were found to show high and efficient photocatalytic reactivity for rather lower concentrations of acetaldehyde molecules. These results clearly indicate that the non-porous TiO<sub>2</sub> nano-powders cannot condense low concentrations of gaseous acetaldehyde molecules on their surfaces in spite of their large surface area (ca. 270 m<sup>2</sup>/g), while the acetaldehyde molecules concentrated within the hydrophobic cavities of the MOR zeolites quickly diffused onto the TiO<sub>2</sub> nano-particles, resulting in the efficient photocatalytic oxidation of acetaldehyde into CO<sub>2</sub> and H<sub>2</sub>O under UV light irradiation. Since the concentration of

gaseous acetaldehyde from home environments is generally at most 50 ppm, TiO<sub>2</sub> nano-particles mechanically blended with MOR zeolite powders are good candidates for the continuous removal of lower and more dilute concentrations of harmful organic compounds.

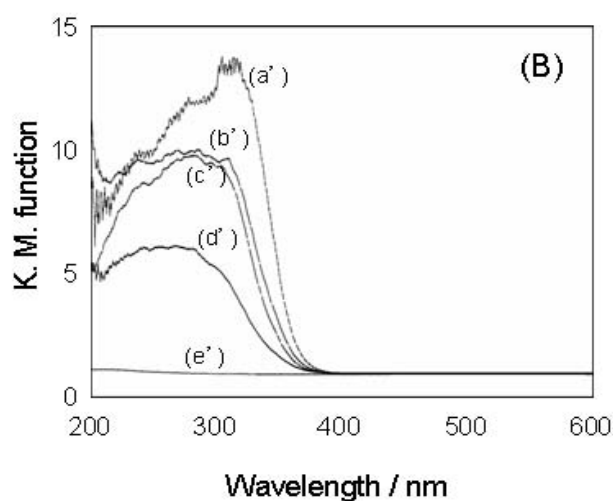
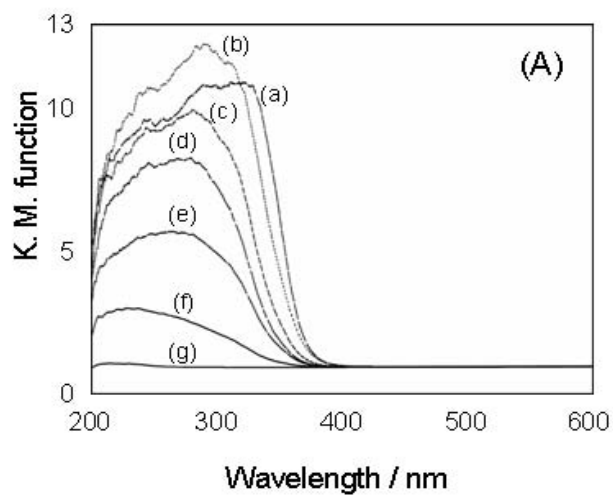
#### **4.4. Conclusions**

The photocatalytic properties of conventional TiO<sub>2</sub> nano-particles (SSP-25, Sakai Chemical Industry Co., Ltd.) could be enhanced by simple mechanical blending with hydrophobic MOR zeolite powders. The optimum amount of the zeolite powders as an adsorbent for the enhancement of the photocatalytic reactivity of the blended TiO<sub>2</sub>/MOR system was estimated to be ca. 80 - 95 wt% since the incident UV light was effectively irradiated onto the entire TiO<sub>2</sub> nano-particles due to the high transparency of the siliceous zeolite powders. Furthermore, the hydrophobic zeolite powders efficiently gathered the gaseous acetaldehyde molecules within their cavities and supplied them onto the TiO<sub>2</sub> surfaces, resulting in an enhancement of the photocatalytic reactivity.



**Figure 4.1.** XRD patterns of TiO<sub>2</sub> nano-particles mechanically blended with MOR (SiO<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> = 1880) zeolite powders.

TiO<sub>2</sub> content (wt%): (a) 100, (b) 50, (c) 20, (d) 10, (e) 5, (f) 1, and (g) 0.

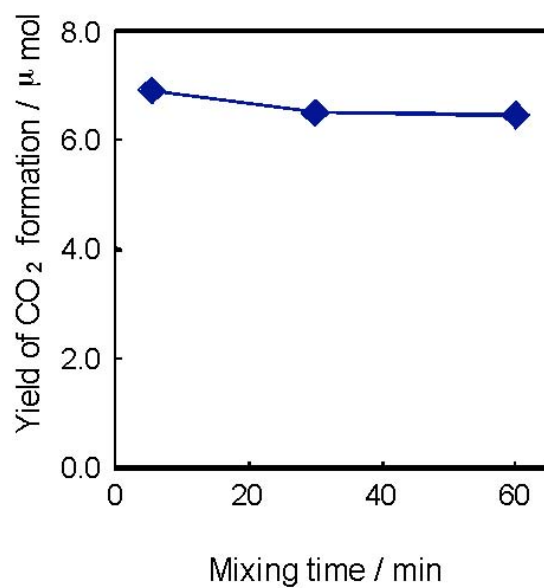


**Figure 4.2-A** Diffuse reflectance UV-Vis absorption spectra of TiO<sub>2</sub> nano-particles mechanically blended with MOR zeolite powders.

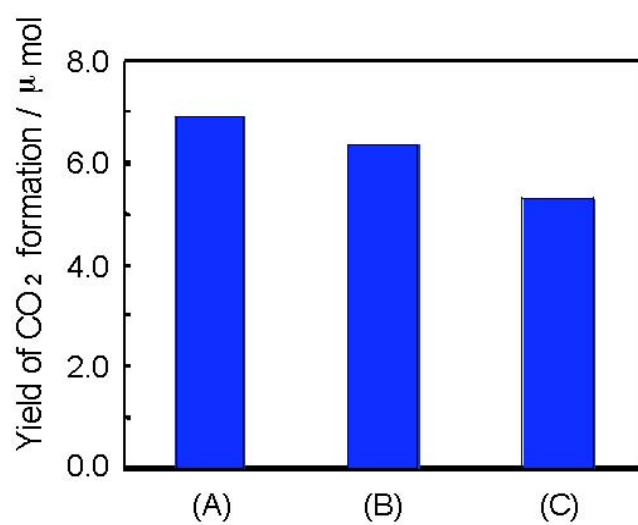
TiO<sub>2</sub> content (wt%): (a) 100, (b) 50, (c) 20, (d) 10, (e) 5, (f) 1, and (g) 0.

**Figure 4.2-B** Diffuse reflectance UV-Vis absorption spectra of TiO<sub>2</sub> nano-particles mechanically blended with non-porous pure SiO<sub>2</sub> powders.

TiO<sub>2</sub> content (wt%): (a') 100, (b') 50, (c') 20, (d') 5, (e) 0.

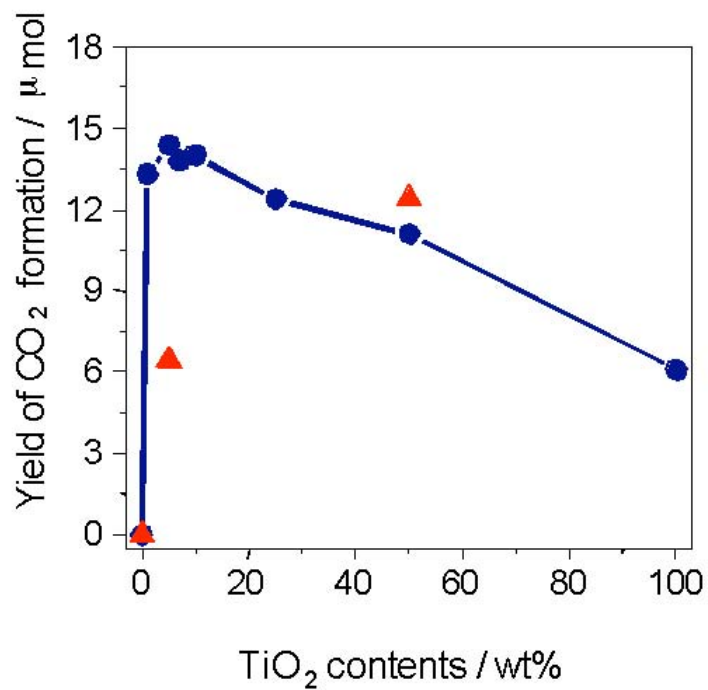


**Figure 4.3** Effect of the mixing time for the TiO<sub>2</sub> and zeolite powders (TiO<sub>2</sub>/MOR ratio = 10/90) in an agate mortar on the photocatalytic reactivity for the oxidation of gaseous acetaldehyde with O<sub>2</sub> under UV light irradiation.

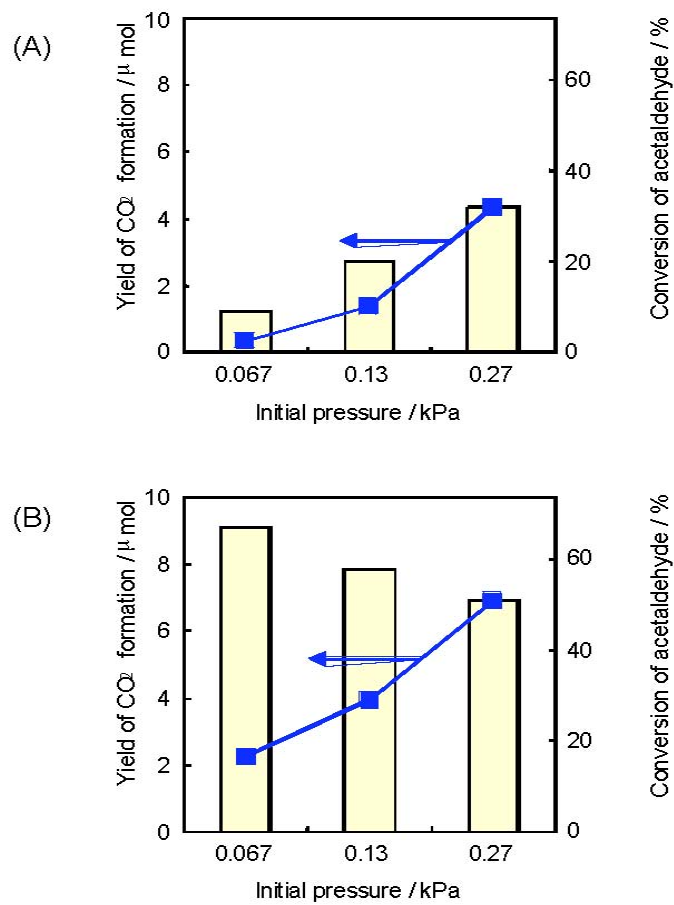


**Figure 4.4** Photocatalytic reactivity (UV light irradiation: 1 h) of the blended TiO<sub>2</sub>/MOR samples (TiO<sub>2</sub>/MOR ratio = 10/90) prepared by: (A) mechanical blending, (B) an ultrasonically aqueous suspension, and (C) without mixing.





**Figure 4.5** Photocatalytic reactivity (UV light irradiation: 3 h) of the TiO<sub>2</sub> nano-particles mechanically blended with MOR and the non-porous SiO<sub>2</sub> powder samples.



**Figure 4.6** Effect of different initial pressures of acetaldehyde on the photocatalytic reactivity of: (A) the TiO<sub>2</sub> nano-particles, and (B) the TiO<sub>2</sub> nano-particles mechanically blended with the MOR zeolite powders (TiO<sub>2</sub>/MOR ratio = 5/95).

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## **Chapter 5**

### **Preparation of the Visible Light Responsive TiO<sub>2</sub> Thin**

### **Film Photocatalysts by a RF-magnetron Sputtering Deposition Method**

## **5. Preparation of Visible Light Responsive TiO<sub>2</sub> Thin Film Photocatalysts by a RF-magnetron Sputtering Deposition Method**

### **5.1. Introduction**

In recent years, TiO<sub>2</sub> photocatalysts have been intensively investigated in various fields and particularly TiO<sub>2</sub> thin films coated on various substrates have shown potential for applications as photofunctional materials not only for their high photocatalytic reactivity but also for their highly wettable properties under UV light irradiation [1-3]. Although various products using TiO<sub>2</sub> thin film photocatalysts have already been commercialized, they do not allow the absorption of visible light and, therefore, necessitates the use of a UV light source. However, in order to realize clean and safe chemical processes as well as the use of abundant solar energy, photocatalysts able to operate even under visible light irradiation are strongly desired. Since an important consideration for widespread and practical applications of high performing TiO<sub>2</sub> photocatalysts is the preparation cost, various methods such as the sol-gel [4-7], chemical vapor deposition (CVD) [8-10], and plasma-enhanced CVD methods [11-13] have been intensively investigated. Among these, the RF (radio frequency) magnetron sputtering deposition method described here was found to be suitable for practical applications since

it enables not only high speed deposition but also deposition of thin films on various substrates with large areas.

We have previously reported on metal ion-implantation into  $\text{TiO}_2$  semiconductor powders and thin films, resulting in an effective modification of their electronic properties to enable the absorption of visible light [14-16]. However, this method necessitates two processes: (i) ionized cluster beam (ICB) deposition to prepare the transparent  $\text{TiO}_2$  thin films; and (ii) modification of the electronic properties of the  $\text{TiO}_2$  semiconductors by a highly advanced metal ion implantation procedure. Since such complexity in the preparation processes impedes mass production at low cost, much easier preparation methods for visible light responsive  $\text{TiO}_2$  thin films are strongly desired in order to realize widespread applications.

In this paper, a more practical alternative preparation process, i. e., a RF magnetron sputtering (RF-MS) deposition method has been successfully applied for the development of transparent  $\text{TiO}_2$  thin films which can induce various significant photocatalytic reactions effectively under UV and visible light irradiation.

## **5.2. Experimental**

$\text{TiO}_2$  thin films were prepared by a RF magnetron sputtering deposition method.

Quartz substrates were ultrasonically cleaned in acetone for 15 min, dried at 373 K for half a day, and then calcined in air at 723 K for 5 h in order to obtain clean surfaces. Before deposition of the TiO<sub>2</sub> thin films, quartz substrates were degassed until less than 5 x 10<sup>-4</sup> Pa (3.8 x 10<sup>-6</sup> Torr) in a high vacuum chamber by a rotary pump and turbo molecular pump. A schematic diagram of this deposition method is shown in Fig. 5.1. The Ar<sup>+</sup> ions in the Ar gas plasma induced by the magnetic and electric fields sputter the TiO<sub>2</sub> target surfaces at high speed to produce sputtered ions such as Ti<sup>4+</sup> and O<sup>2-</sup>. These ions produced by the Ar gas plasma are accumulated onto the substrate surfaces to form TiO<sub>2</sub> thin films. In a conventional reactive sputtering method, oxide thin films are prepared by sputtering a metallic Ti target in the presence of O<sub>2</sub> atmosphere as the reactive gas. However, in this study, since a TiO<sub>2</sub> plate with a rutile structure (High Purity Chemicals Lab. Corp., grade: 99.99 %) was applied as the ion source material (sputtering target), only Ar was used as the sputtering gas without coexisting O<sub>2</sub> as the reactive gas. The physicochemical, mechanical and photocatalytic properties of the obtained TiO<sub>2</sub> thin films were strongly affected by the preparation conditions such as induced RF power, substrate temperature, distance between target and substrates (D<sub>TS</sub>), and sputtering gas flow rate, etc. The induced RF power was adjusted to 300 W, the substrate temperatures were changed from 373 K to 973 K and D<sub>TS</sub> was set at 80 mm. The flow rate of Ar gas was kept at 25 SCCM and the



pressure of the sputtering gas under deposition process was about 2.0 Pa ( $1.5 \times 10^{-3}$  Torr). Since the TiO<sub>2</sub> thin films are deposited under high vacuum conditions, contamination with some impurities into TiO<sub>2</sub> thin films could be avoided. No post-calcination treatments were carried out. The film thicknesses of these thin films were controlled at ca. 1  $\mu\text{m}$  by changing the deposition time.

Characterization of these TiO<sub>2</sub> thin films were carried out by XRD (Rigaku, RINT-1200) and UV-Vis absorption measurements (Shimadzu, UV-2200A). The surface morphologies of the thin films were investigated by SEM (Hitachi, S-4700) and AFM (Seiko Instruments, SPA300) analyses. Furthermore, their atomic compositions from top surface to deep bulk were investigated by AES measurements (ULVAC-PHI, SAM670).

The photocatalytic reactivities of the TiO<sub>2</sub> thin films were evaluated by carrying out the decomposition of NO under UV ( $\lambda > 270$  nm) or visible ( $\lambda > 450$  nm) light irradiation [14, 17]. Light irradiation was carried out using a conventional 100 W high-pressure Hg lamp (Toshiba, SHL-100UVQ-2) through cutoff (Toshiba Glass, UV-27 or Y-45) and water filters at 275 K. The reaction products were analyzed by a gas chromatograph equipped with a TCD detector.

### **5.3. Results and Discussion**

The optical property is one of the most important factors determining the photocatalytic performance. Figure 5.2 shows the UV-Vis absorption (transmittance) spectra of the TiO<sub>2</sub> thin films prepared at different preparation temperatures. The TiO<sub>2</sub> thin films prepared at low temperatures ( $T < 473$  K) showed high transparency and clear interference fringes in the visible light region, like TiO<sub>2</sub> thin films prepared by sol-gel [4-7] or ionized cluster beam (ICB) deposition [14, 17]. These results clearly indicate that stoichiometric and uniform TiO<sub>2</sub> thin films can be prepared by using a TiO<sub>2</sub> plate as the sputtering target and Ar as the sputtering gas without the coexistence of O<sub>2</sub> as the reactive gas. As the preparation temperatures increased, the TiO<sub>2</sub> thin films were found to show effective absorption in visible light regions with a maximum for the thin film deposited at 873 K. Since the amount of impurities included in the TiO<sub>2</sub> target material is very low (lower than 0.1 %), hardly any impurities were included in the deposited thin films. The TiO<sub>2</sub> thin films prepared by the sputtering TiO<sub>2</sub> target in the presence of O<sub>2</sub> at 873 K did not exhibit any significant absorption in visible light regions (data not shown). These results indicate that visible light-responsive TiO<sub>2</sub> thin films could successfully be prepared only when the TiO<sub>2</sub> target as the ion source is sputtered with Ar gas without the coexistence of O<sub>2</sub> gas at relatively high temperatures ( $T > 773$  K).

The XRD patterns of the TiO<sub>2</sub> thin films prepared at different substrate temperatures

are shown in Fig. 5.3. The TiO<sub>2</sub> thin films prepared at temperatures lower than 673 K showed a specific diffraction peak at around 37.9 degrees attributed to the anatase (004) phase. Since a conventional TiO<sub>2</sub> powder (Degussa, P-25) does not show such an intense peak due to the (004) phase of the anatase structure, this (004) phase may be considered characteristic to the TiO<sub>2</sub> thin films prepared by this RF-MS deposition method. For the TiO<sub>2</sub> thin films prepared at 873 K, a diffraction peak at 25.4 degrees due to the anatase (101) phase was found to largely decrease, however, the (004) phase was relatively stable. The TiO<sub>2</sub> thin films readily able to absorb visible light were mainly composed of a rutile structure of for the TiO<sub>2</sub>. As summarized in Table 5.1, the primary particle sizes of the anatase and rutile phases in the TiO<sub>2</sub> thin films were determined by using the Scherrer's equation from the diffraction peaks at ca. 25 and 28 degrees, respectively. The particle sizes of the TiO<sub>2</sub> which make up the thin films were not largely affected by the preparation temperatures and were constant at about 20 nm. Since thin films are generally strained by interfacial stress on the substrates, crystallization along their depth direction may be restricted.

The TiO<sub>2</sub> thin films prepared by the RF-MS deposition method were found to efficiently induce the photocatalytic decomposition of NO into N<sub>2</sub> and N<sub>2</sub>O under UV light ( $\lambda > 270$  nm) irradiation. Figure 5.4 shows the reaction time profiles for the

photocatalytic decomposition of NO over the TiO<sub>2</sub> thin film prepared at 473 K and a commercial TiO<sub>2</sub> powder (Degussa, P-25) under UV light irradiation. The transparent TiO<sub>2</sub> thin films prepared by the RF-MS method showed almost the same photocatalytic reactivity for the reductive decomposition of NO under UV light irradiation as TiO<sub>2</sub> powders known to be highly reactive photocatalysts. Figure 5.5 shows the effect of the preparation temperatures on the photocatalytic reactivity of the TiO<sub>2</sub> thin films for the decomposition of NO under UV light irradiation. The TiO<sub>2</sub> thin film prepared at 473 K showed the highest photocatalytic performance under UV light irradiation, however, as the preparation temperatures increased, the reactivity of the TiO<sub>2</sub> thin films decreased. These results can be explained by the observation that the TiO<sub>2</sub> thin films prepared at relatively lower temperatures are highly transparent and, thus, the incident UV light can pass through the thin films, resulting in effective utilization of the incident light to generate electron-hole pairs. However, since the thin film sample prepared at 373 K also included some amorphous phase which means a low crystallinity, this sample showed lower reactivity as compared to that prepared at 473 K. XRD measurements also revealed that the TiO<sub>2</sub> thin films prepared at lower temperatures mainly consist of an anatase structure, on the other hand, those prepared at relatively higher temperatures crystallize well to a rutile structure. These differences in the crystal structures of the TiO<sub>2</sub> thin films showed a

good correspondence with their photocatalytic reactivities under UV light irradiation.

The photocatalytic reactivities of these TiO<sub>2</sub> thin films under visible light irradiation ( $\lambda > 450$  nm) were also investigated. As shown in Fig. 5.6, although the reference TiO<sub>2</sub> powder (Degussa, P-25) did not show any photocatalytic reactivity under visible light irradiation, the TiO<sub>2</sub> thin film photocatalyst prepared at 873 K was found to efficiently decompose NO into N<sub>2</sub> and N<sub>2</sub>O even under visible light irradiation. Figure 5.7 shows the effects of the preparation temperatures on the photocatalytic reactivity under visible light irradiation as well as on the relative intensities at wavelengths of 450 nm in the UV-Vis absorption spectra, as shown in Fig. 5.2. As the preparation temperatures increased, the photocatalytic reactivity under visible light irradiation was found to increase as well, reaching a maximum at 873 K. However, the thin film sample prepared at 373 K showed rather low reactivity under visible light irradiation. Since this sample prepared at 373 K showed lower crystallinity as compared to that prepared at 473 K, some oxygen vacant sites may exist on the surface of the thin film samples and work as NO reductive sites. Moreover, the order of the photocatalytic reactivity under visible light irradiation corresponds well with the relative intensities at wavelengths of 450 nm in the UV-Vis absorption spectra. These results clearly indicate that the TiO<sub>2</sub> thin films prepared by a RF-MS method do, in fact, work as photocatalysts even under visible light irradiation of

wavelengths longer than 450 nm. However, since the semiconducting powder samples which absorb visible light do not always show photocatalytic reactivity under visible light irradiation, other factors to determine the photocatalytic performance, such as BET surface areas, surface roughness, oxygen vacancies which may work as recombination centers, also need to be investigated.

TiO<sub>2</sub> photocatalysts are also known to show the potential to completely oxidize various organic compounds into harmless CO<sub>2</sub> and H<sub>2</sub>O under UV light irradiation [18-21]. These visible light-responsive TiO<sub>2</sub> thin films prepared by the RF-MS method were confirmed to be effective for other photocatalytic reactions as such as the complete oxidation of various organic compounds into CO<sub>2</sub> and H<sub>2</sub>O as well as the splitting of H<sub>2</sub>O into H<sub>2</sub> and O<sub>2</sub> under UV ( $\lambda > 270$  nm) and visible light ( $\lambda > 450$  nm) irradiation [22-25].

The mechanisms for the efficient absorption of visible light has been discussed from the viewpoint of the surface morphology and O/Ti ratios of the TiO<sub>2</sub> thin films prepared by the RF-MS method. Figure 5.8 shows the AFM images of the TiO<sub>2</sub> thin films prepared at 473 K and 873 K. The surface morphology and roughness of the films prepared at 873 K was found to be about three times higher than the films prepared at 473 K. Additionally, cross-sectional SEM images of the TiO<sub>2</sub> thin films prepared at 473 K and 873 K are displayed in Fig. 5.9. Top views of the SEM images for these thin films were almost the

same as the above AFM images. Clear differences between the UV and visible light responsive TiO<sub>2</sub> thin films could be observed in their cross sectional views. The film prepared at 473 K has a structure in which nano-sized TiO<sub>2</sub> particles randomly sintered with each other. On the other hand, the films prepared at 873 K were found to possess a unique and characteristic structure in which TiO<sub>2</sub> single crystals with a columnar structure (diameter: ca. 100 nm) are orderly aligned. This unique structure could be observed only when the TiO<sub>2</sub> thin films were prepared at relatively higher temperatures by the RF-MS method. From these investigations, the high transparency of these TiO<sub>2</sub> thin films were found to be associated with the construction of an assembly of closely packed TiO<sub>2</sub> particles with small surface roughness. On the other hand, the efficient absorption of visible light may be associated with their characteristic structures, such as the orderly aligned columnar TiO<sub>2</sub> crystals, obtained only by the RF-MS method. For further studies on the mechanism of the absorption of visible light, depth profiles of the O/Ti atomic ratio were investigated by AES measurements. As shown in Fig. 5.10, the O/Ti atomic ratio of the transparent TiO<sub>2</sub> thin film prepared at 473 K was constant at 2.0 from the surface to deep bulk, suggesting a stoichiometric TiO<sub>2</sub> composition. On the other hand, the O/Ti atomic ratio of the visible light-responsive TiO<sub>2</sub> thin film prepared at 873 K was found to decrease gradually from surface to deep bulk reaching about 1.933. Such a declined

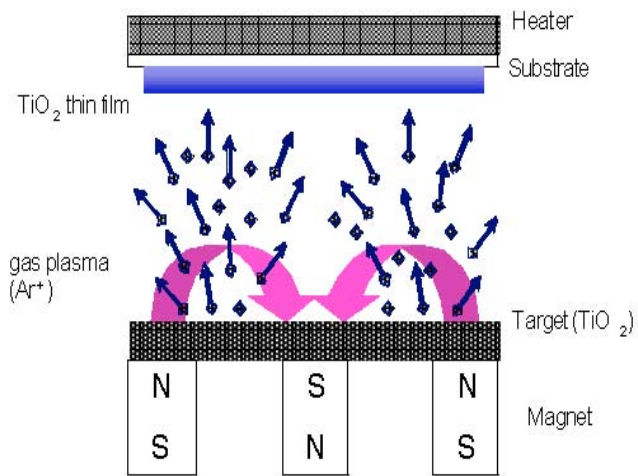
structure in the O/Ti atomic ratio for the visible light-responsive TiO<sub>2</sub> thin films was confirmed to be stable even after calcination at 773 K, since the stoichiometric TiO<sub>2</sub> layer at the surface (thickness; ca. 100 nm) worked as a passive layer to protect the bulk sections. Moreover, the declined structure may be closely associated with the modification of the electronic properties of the TiO<sub>2</sub> semiconductor, enabling the efficient absorption of visible light. It has already been reported that small amounts of oxygen vacancies in the TiO<sub>2</sub> lattice give rise to the distortion of the TiO<sub>2</sub> octahedral unit and weaken the Ti-O bonds, resulting in a reduction of the splitting between the bonding and nonbonding levels [26]. However, the further investigations on the role of the oxygen vacancy for the absorption of visible light are yet necessary. Taking these results into consideration, such a characteristic declined composition of the TiO<sub>2</sub> thin films may be the determining factor in modifying the electronic properties so that they are able to operate as efficient photocatalysts even under visible light irradiation.

#### **5.4. Conclusions**

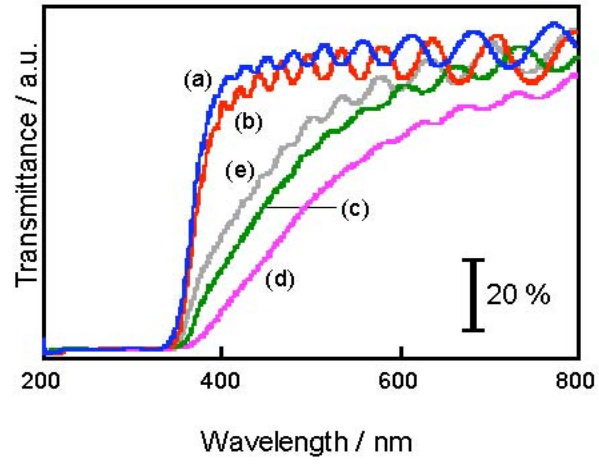
Transparent TiO<sub>2</sub> thin films which operate efficiently as a photocatalyst under UV and visible light irradiation were successfully developed using a single process by controlling



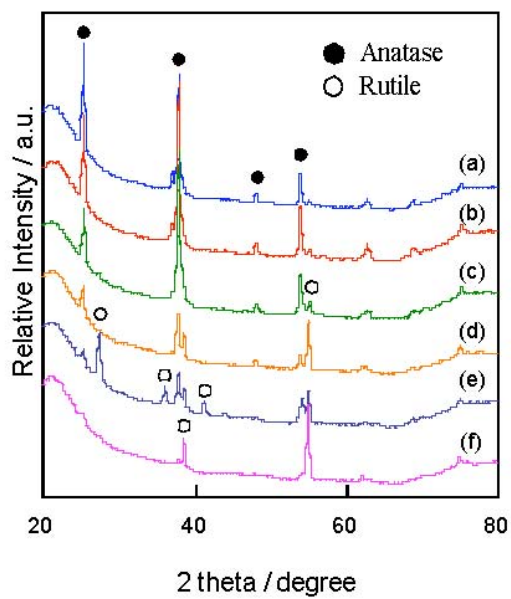
the preparation conditions such as the target material, sputtering gas and preparation temperatures in a RF-magnetron sputtering (RF-MS) deposition method. The TiO<sub>2</sub> thin films prepared at relatively low temperatures ( $T < 473$  K) showed high transparency and effective photocatalytic reactivity for the decomposition of NO as well as the oxidation of various organic compounds under UV light irradiation. On the other hand, the thin films prepared at relatively high temperatures ( $T > 773$  K) exhibited a pale yellow color, indicating that these thin films can absorb visible light efficiently so that much higher photocatalytic reactivity for the above reactions were possible even under visible light irradiation. From various characterization studies of these thin films, the unique structure of the orderly aligned columnar TiO<sub>2</sub> single crystals could be observed along with a declined structure in which the O/Ti atomic ratio slightly decreased from surface to deep bulk. These characteristic structural factors are considered to play an important role in the efficient modification of the electronic properties of these TiO<sub>2</sub> semiconductors, enabling the absorption of visible light. These findings are expected to open the way to new possibilities in the widespread and cost-efficient production of visible light-responsive TiO<sub>2</sub> photocatalysts that can utilize safe and clean solar energy for beneficial purposes.



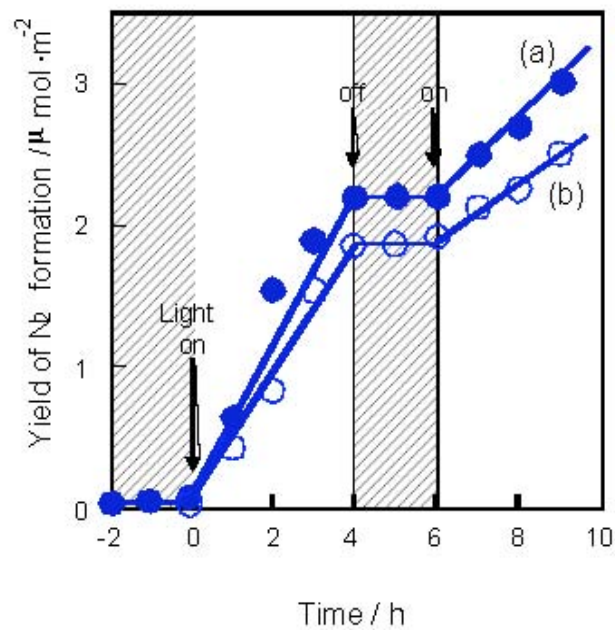
**Figure 5.1** Schematic diagram of the RF magnetron sputtering (RF-MS) deposition method.



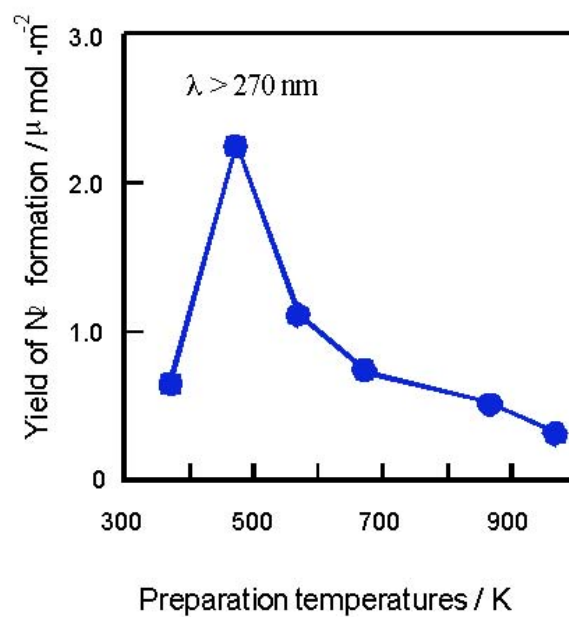
**Figure 5.2** UV-Vis absorption spectra of the TiO<sub>2</sub> thin films prepared at: (a) 373, (b) 473, (c) 673, (d) 873 and (e) 973 K.



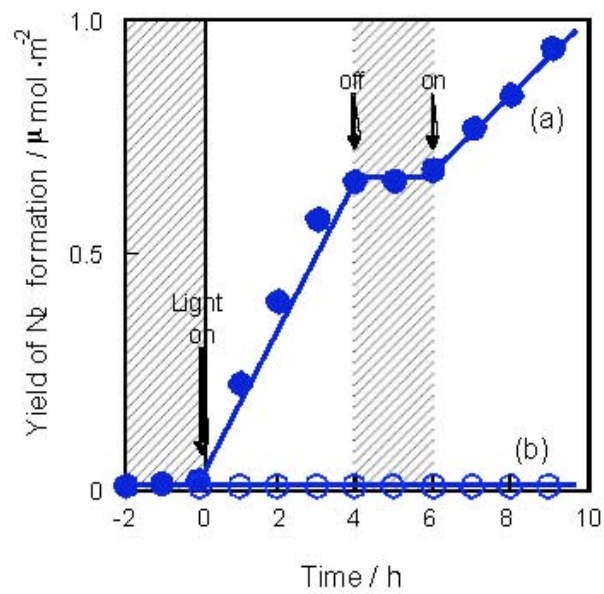
**Figure 5.3** XRD patterns of the TiO<sub>2</sub> thin films prepared at: (a) 473, (b) 573, (c) 673, (d) 773, (e) 873 and (f) 973 K.



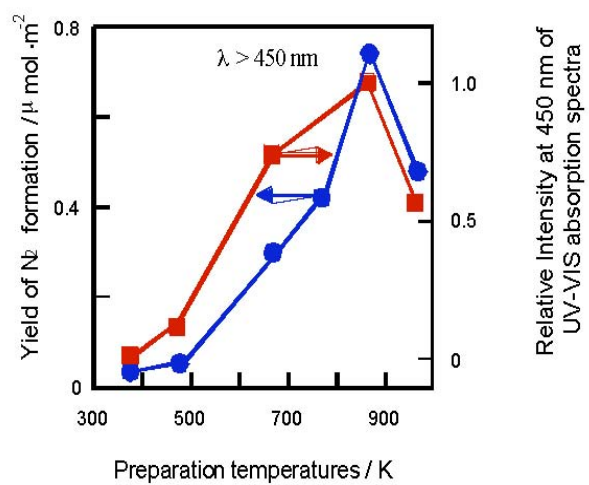
**Figure 5.4** Reaction time profiles of the photocatalytic decomposition of NO on: (a) the transparent TiO<sub>2</sub> thin film prepared at 473 K; and (b) the commercial TiO<sub>2</sub> powder (Degussa, P-25) under UV light ( $\lambda > 270$  nm) irradiation at 275 K.



**Figure 5.5** Effect of the preparation temperatures on the photocatalytic reactivity of the TiO<sub>2</sub> thin films for the decomposition of NO under UV light ( $\lambda > 270 \text{ nm}$ ) irradiation.

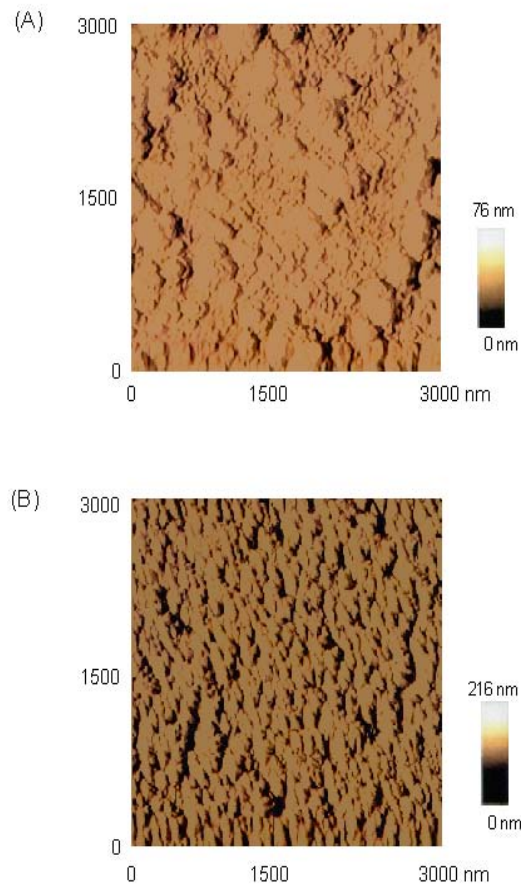


**Figure 5.6** Reaction time profiles of the photocatalytic decomposition of NO on: (a) the visible light responsive  $\text{TiO}_2$  thin film prepared at 873 K; and (b) the commercial  $\text{TiO}_2$  powder (Degussa, P-25) under visible light ( $\lambda > 450 \text{ nm}$ ) irradiation at 275 K.

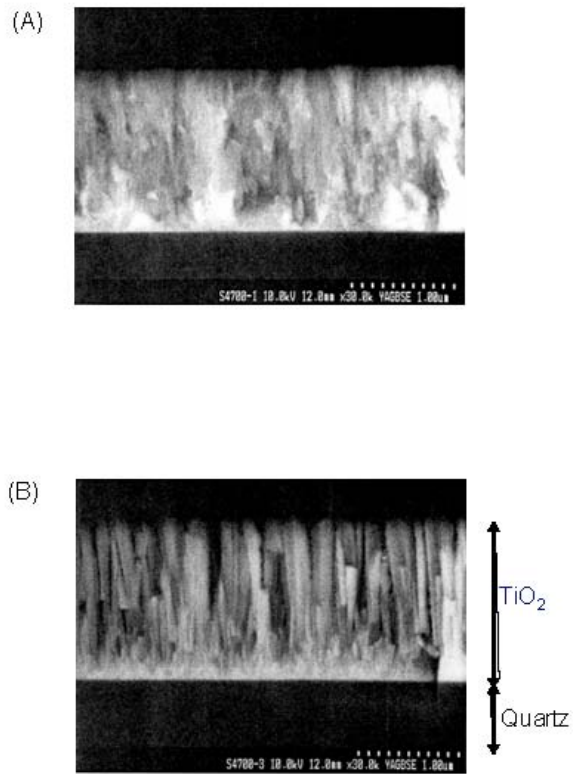


**Figure 5.7** Effect of the preparation temperatures on the photocatalytic reactivity for the decomposition of NO under visible light ( $\lambda > 450$  nm) irradiation as well as the relative intensities at 450 nm in the UV-Vis absorption spectra.

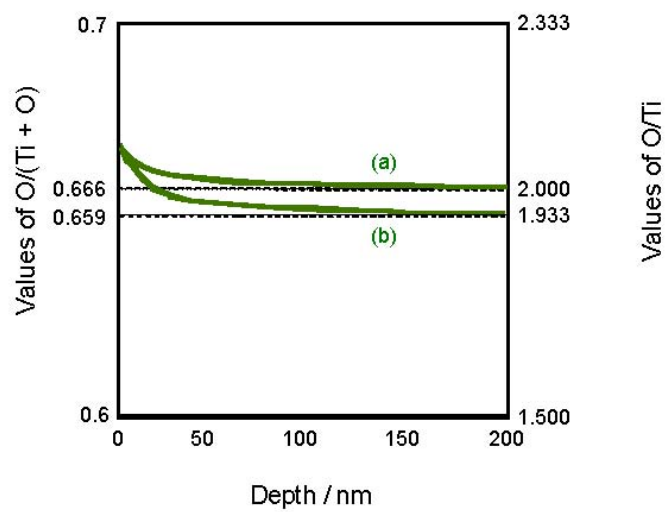




**Figure 5.8** AFM images of the  $\text{TiO}_2$  thin film prepared at (A) 473 K and (B) 873 K.



**Figure 5.9** Cross-sectional SEM images of the TiO<sub>2</sub> thin film prepared at (A) 473 K and (B) 873 K.



**Figure 5.10** Depth profiles of the O/(Ti+O) and O/Ti ratios as determined by AES measurements for the TiO<sub>2</sub> thin films prepared at (a) 473 K and (b) 873 K.

Preparation temperature (K)	Particle size (nm)	Crystal phase
473	18.4	Anatase
573	16.5	Anatase
673	18.9	Anatase
773	21.4	Anatase
873	20.8	Rutile (mainly)
973	---	Rutile

Scherrer's equation:  $D = K \lambda / \beta \cos \theta$

$D$  : particle size (nm)

$\lambda$  : wavelength of CuK $\alpha$  (0.15405 nm)

$\beta$  : half width (rad)

$\theta$  : Bragg angle (degree)

$K$  : constant (in this case: 0.9)

**Table 5.1.** Particle sizes determined by the results of XRD measurements.

## 5.5. References

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## **Chapter 6**

### **General Conclusions**



## **6. General Conclusions**

In this thesis, the study of the preparations of TiO<sub>2</sub> nano-particle photocatalysts by a multi-gelation method and, suppressed recombination of electrons and holes and its role in the improvement of the photoreactivity of flame-synthesized TiO<sub>2</sub> nanopowders, enhancement of the photocatalytic reactivity of TiO<sub>2</sub> nano-particles by a simple mechanical blending with hydrophobic MOR zeolites, and preparation of the visible light responsive TiO<sub>2</sub> thin film photocatalysts by a RF-magnetron sputtering deposition method have been investigated.

The main results obtained have been summarized below.

## **Summary of Chapter2**

Chapter2 deals with the synthesis, characterization and photocatalytic performance of TiO<sub>2</sub> nanopowders prepared by a flame-synthesis method. The Anatase phase-rich TiO<sub>2</sub> nanopowders containing small amounts of the rutile phase were fabricated by the flame method. The photoexcited states of the TiO<sub>2</sub> nanopowders were directly determined by in-situ NEXAFS measurements under UV light irradiation. The present findings give direct experimental evidence by the in-situ NEXAFS showing, for the first time, that the electrons trapped in the anatase/rutile grain boundary suppress the recombination of electrons and holes, and in turn, this suppression of recombination directly contributes to an improvement in the photocatalytic reactivities for the decomposition of 2-propanol.

### **Summary of Chapter3**

Chapter3 deals with the TiO<sub>2</sub> photocatalysts prepared by a multi-gelation method and the effect of the changes in the pH during the pH swing times, i.e., by a controlled pH swing, on the morphology of the TiO<sub>2</sub> particles. TiO<sub>2</sub> photocatalysts prepared by a controlled pH swing method could retain the anatase phase even at calcination temperatures of 750 °C at higher pH swing numbers. However, other important parameters such as particle size, surface area, pore volume, pore size as well as the anatase/rutile phase ratio could not be controlled well by this method. The addition of HCl acid during preparation showed detrimental effects on the morphology of the particles. On the other hand, TiO<sub>2</sub> catalysts prepared by an uncontrolled pH swing method showed better performance, especially in control of such important parameters. The results of 2-propanol oxidation showed that control of the anatase/rutile ratio, the pore volume as well as pore diameter of the TiO<sub>2</sub> nano-particles are important factors in realizing the efficient photocatalytic degradation of organic compounds.

## Summary of Chapter4

Chapter4 deals with the photocatalytic oxidation of gaseous acetaldehyde with  $O_2$  on commercial  $TiO_2$  nano-particles. The photocatalytic properties of conventional  $TiO_2$  nano-particles (SSP-25, Sakai Chemical Industry Co., Ltd.) could be enhanced by the simple method of mechanical blending with hydrophobic MOR zeolite powders. The optimum amount of the zeolite powders as an adsorbent for the enhancement of the photocatalytic reactivity of the blended  $TiO_2$ /MOR system was estimated to be ca. 80 - 95 wt% since the incident UV light was effectively irradiated onto the whole part of the  $TiO_2$  nano-particles due to the high transparency of the siliceous zeolite powders. Furthermore, the hydrophobic zeolite powders efficiently gathered the gaseous acetaldehyde molecules within their cavities and supplied them onto the  $TiO_2$  surfaces, resulting in the enhancement of the photocatalytic reactivity.

## Summary of Chapter5

Chapter5 deals with the  $\text{TiO}_2$  thin film photocatalysts which could induce photoreactions under visible light irradiation. Transparent  $\text{TiO}_2$  thin films which operate efficiently as a photocatalyst under UV and visible light irradiation were successfully developed using a single process by controlling the preparation conditions such as the target material, sputtering gas and preparation temperatures in a RF-magnetron sputtering (RF-MS) deposition method. The  $\text{TiO}_2$  thin films prepared at relatively low temperatures ( $T < 473 \text{ K}$ ) showed high transparency and effective photocatalytic reactivity for the decomposition of NO as well as the oxidation of various organic compounds under UV light irradiation. On the other hands, the thin films prepared at relatively high temperatures ( $T > 773 \text{ K}$ ) exhibited a pale yellow color, indicating that these thin films can absorb visible light efficiently so that much higher photocatalytic reactivity for above reactions were possible even under visible light irradiation. From various characterization studies of the visible light responsive  $\text{TiO}_2$  thin films, the unique structure of orderly aligned columnar  $\text{TiO}_2$  single crystals could be observed along with a declined structure in which the O/Ti atomic ratio slightly decreased from surface to deep bulk. These characteristic structural factors might play an important role in the efficient

modification of electronic properties of TiO<sub>2</sub> semiconductors, enabling the absorption of visible light. These findings are expected to open the way to new possibilities in the widespread and affordable production of the visible light responsive TiO<sub>2</sub> photocatalysts that can utilize safe and clean solar energy.

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2010, February  
Sakai, Osaka

## LIST OF PUBLICATIONS

1. New Trends in the Nanoscience and Nanotechnology of  
Titanium Oxide-based Photocatalysts as an Environmentally-friendly Catalyst

Masato Takeuchi, Shirou Sakai, Afshin Ebrahimi, Masaya Matsuoka,  
Masakazu Anpo

*Proc. in Nanosci. Nanotec. Conf.*, 5-10 (Turkey, Istanbul, 2008).

2. Suppressed Recombination of Electrons and Holes and its role the improvement of  
Photoreactivity of Flame-synthesized TiO<sub>2</sub> Nanopowders

Hoon Park, Hyunseock Jie, Keun-Hwa Chae, Shinya Higashimoto,

Shirou Sakai, Masakazu Anpo, Jong-Ku Park, Dok-Yol Lee

submitted to *Res.Chem.Intermed.*, (2010)

3. Preparation of TiO<sub>2</sub> nano-particle photocatalysts by a multi-gelation method: The  
effect of pH change

Bernaardshow Neppolian, Diana Eddy Rakhmawaty, Shiro Sakai,

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4. Enhancement of the photocatalytic reactivity of TiO<sub>2</sub> nano-particles by a simple mechanical blending with hydrophobic mordenite (MOR) zeolite

Masato Takeuchi , Junichi Deguchi , Manabu Hidaka , Shiro Sakai , Kyoungja Woob,  
Pyuck-Pa Choi , Jong-Ku Park , Masakazu Anpo  
*Applied Catalysis B: Environmental*, 89, 406-410 (2009).

5. Preparation of Visible Light-Responsive TiO<sub>2</sub> Thin Film Photocatalysts by a RF-magnetron Sputtering Deposition Method

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6. Application of Highly Functional Ti-oxide-based Photocatalysts in Clean Technologies

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*Topics in Catalysis*, 52, 1651-1659 (2009).

本論文の基礎となる発表論文

No.	論文題目	著者名	発表誌名	本論文との対応
1	New Trends in the Nanoscience and Nanotechnology of Titanium Oxide-based Photocatalysts as an Environmentally-friendly Catalyst	M. Takeuchi S. Sakai A. Ebrahimi M. Matsuoka M. Anpo	Proc. Nanosci. Nanotech. Conf., pp. 5-10 (Turkey, Istanbul, 2008).	第 1 章
2	Suppressed Recombination of Electrons and Holes and its Role for the Improvement of Photoreactivity of Flame-synthesized TiO <sub>2</sub> Nanopowders	H. Park H. Jie K.-H. Chae S. Higashimoto S. Sakai M. Anpo J.-K. Park D.-Y. Lee	submitted to Res. Chem. Intermed.	第 2 章
3	Preparation of TiO <sub>2</sub> Nano-particle Photocatalysts by a Multi-gelation Method: The Effect of pH Change	B. Neppolian D. Rakhmawaty S. Sakai Y. Okada H. Nishijima M. Anpo	Res. Chem. Intermed., Vol. 34, pp. 103-111 (2008).	第 3 章
4	Enhancement of Photocatalytic Reactivity of TiO <sub>2</sub> Nano-particles by a Simple Mechanical Blending with Hydrophobic Mordenite (MOR) Zeolite	M. Takeuchi J. Deguchi M. Hidaka S. Sakai K. Woo P.-P. Choi J.-K. Park M. Anpo	Appl. Catal. B: Environmental, Vol. 89, pp. 406-410 (2009).	第 4 章
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